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Investigation and Manipulation of Domains on Small Length Scales in Ferroelectric Materials: Limits and Opportunities

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Oak Ridge National Laboratory

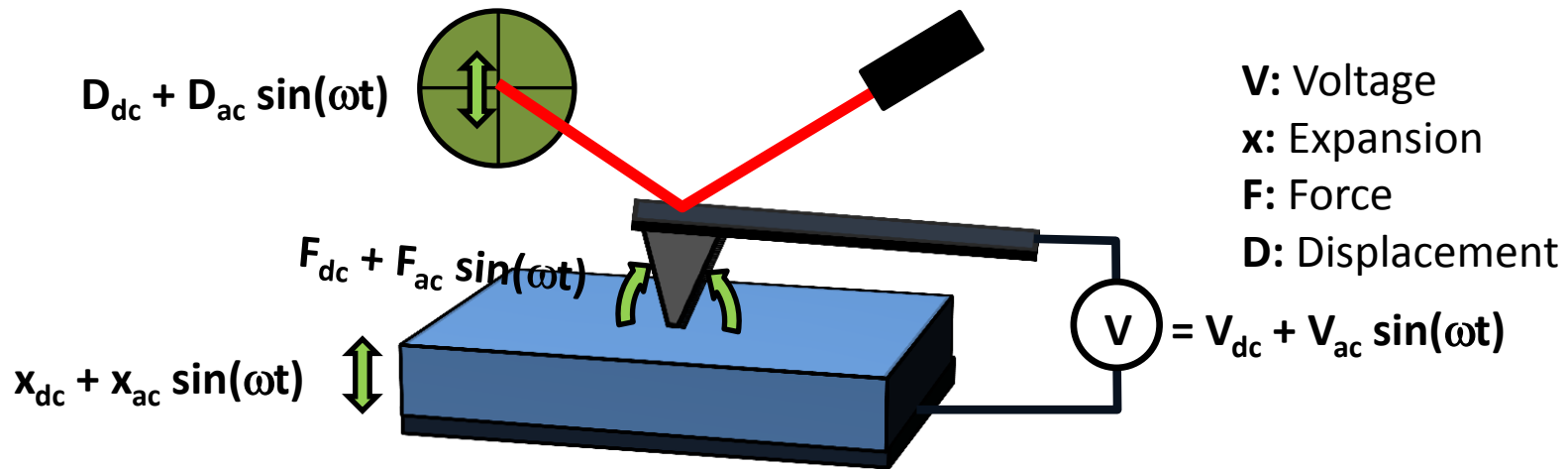
Center for Nanophase Materials Sciences

- What can we study?
 - *Electromechanical phenomena overview*
 - *Ferroelectrics in particular*
 - *“New” ferroelectrics (ultra-thin, strained, ...)*
- Dominant signal contribution for ferroelectrics
 - *d_{33} and electrostatic tip-sample interaction*
- Ionic transport in Li-ion battery cathodes
- Solid-liquid interface and EDL for ionic liquids
- Summary

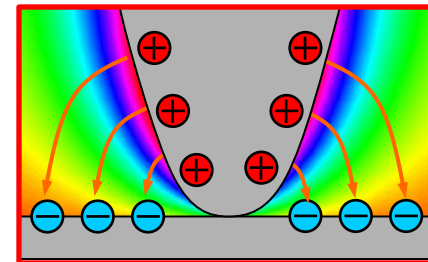
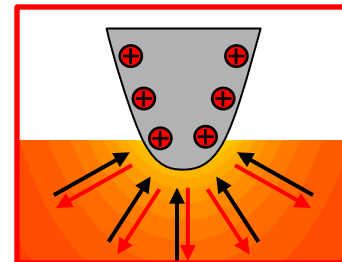
Scanning Probe Microscopy (SPM)

Investigate electromechanical phenomena on the nanoscale

Practical: Apply a bias \rightarrow Measure change in cantilever position



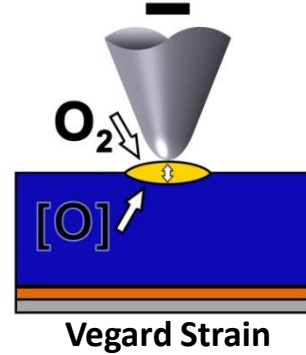
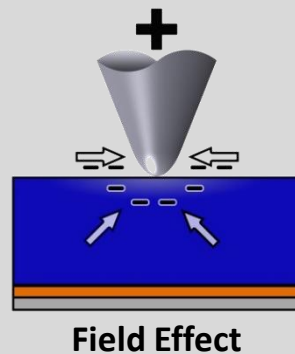
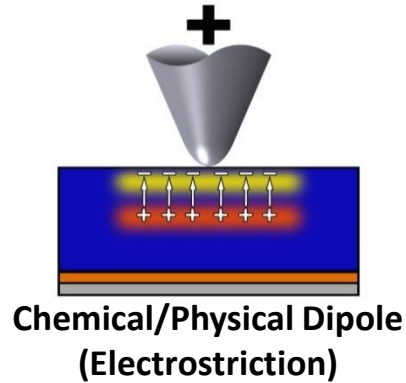
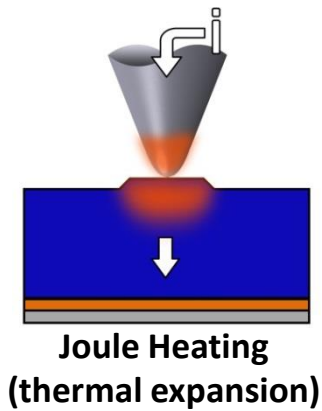
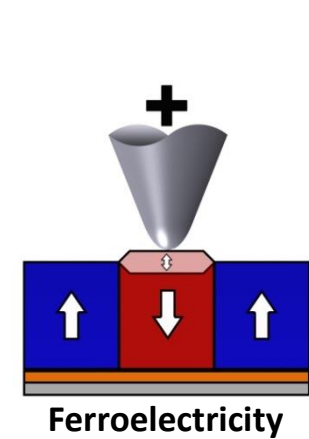
- 1) Sample volume expansion vs. force on tip
- 2) Static vs. dynamic



Physical

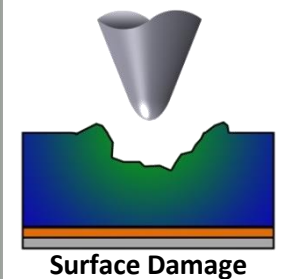
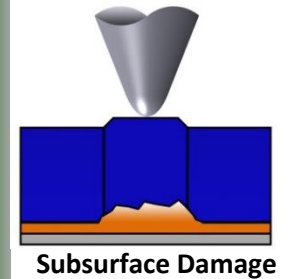
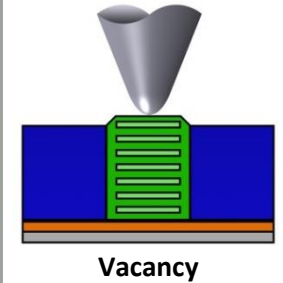
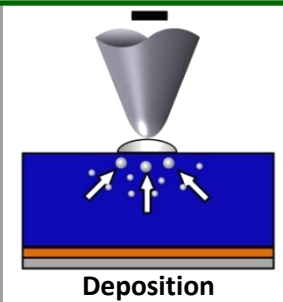
Electrochemical

Reversible Reactions



electrostatic
force on tip

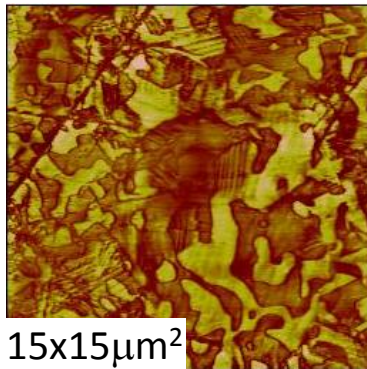
Irreversible Reactions



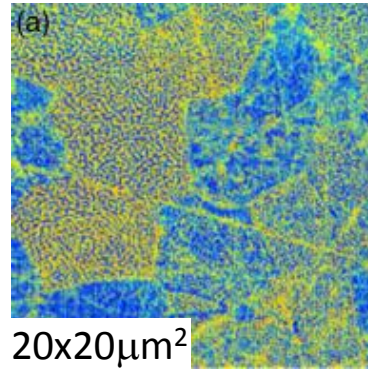
Piezoresponse Force Microscopy (PFM): Domain imaging (V_{ac}) and manipulation (V_{dc})

Imaging of OP and IP domains in ceramics and thin films

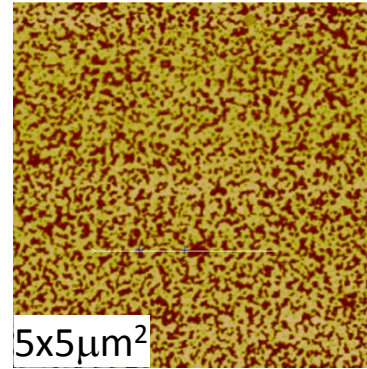
Ceramic PZT



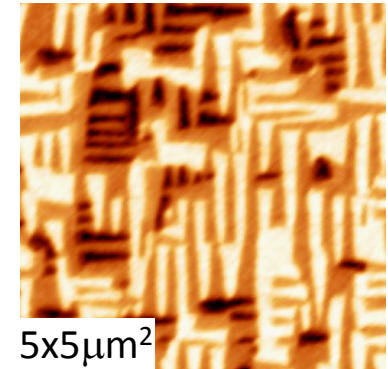
Ceramic PLZT



PZT thin film

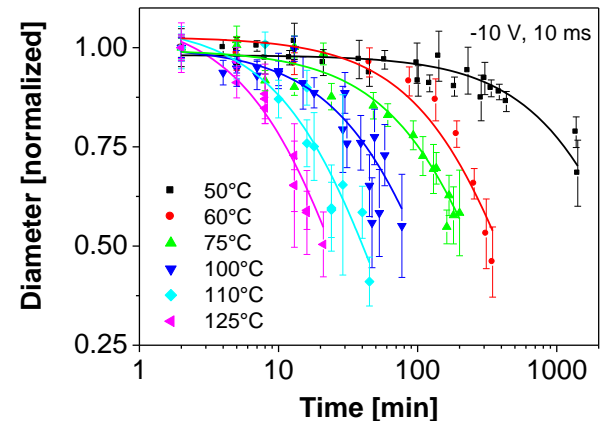
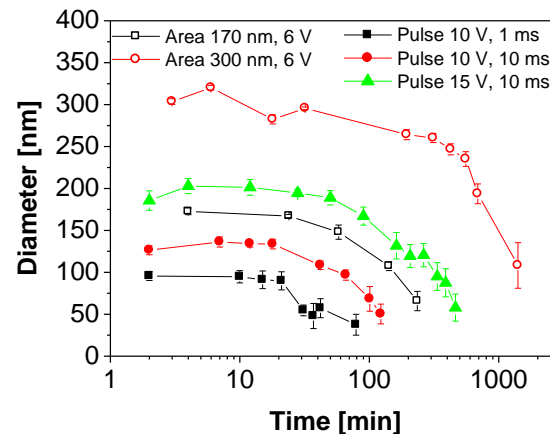


BFO thin film



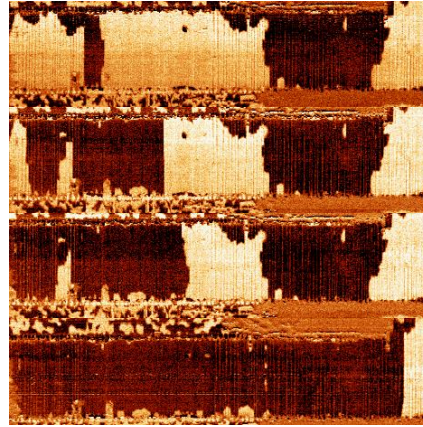
Stability of nanodomains

PZT thin film

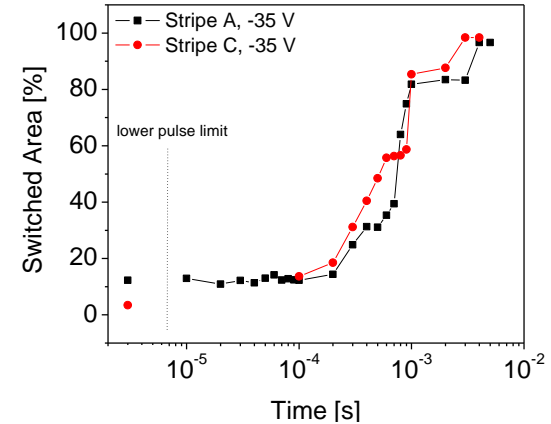


In-plane capacitor switching

BFO thin film



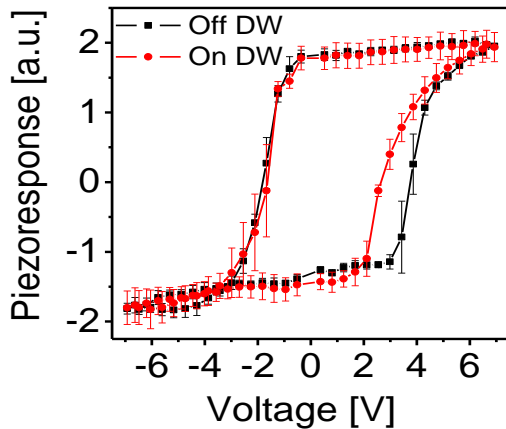
Increasing time



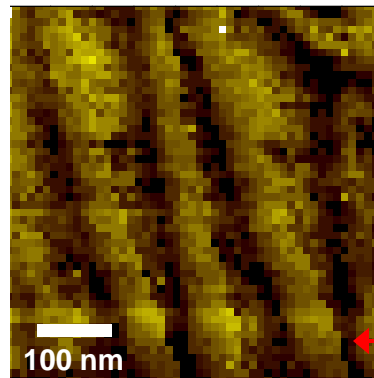
Balke et al., *Adv. Func. Mater.* 20, 3466 (2010)

Map of switching parameters

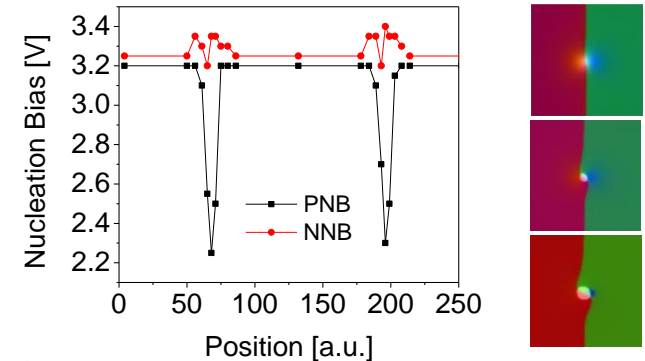
BFO thin film



Map of positive nucleation voltage



Simulation of nucleation voltages

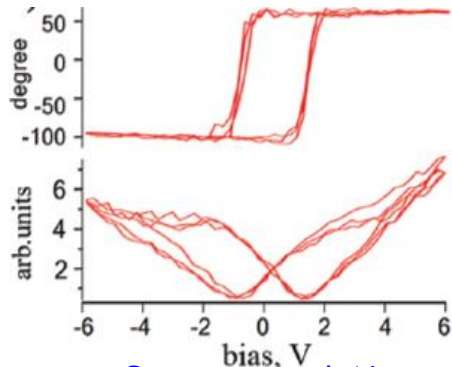


Balke et al., *Nat. Nanotechnol.* 4, 868 (2009)

Pushing the limits ... is everything ferroelectric?

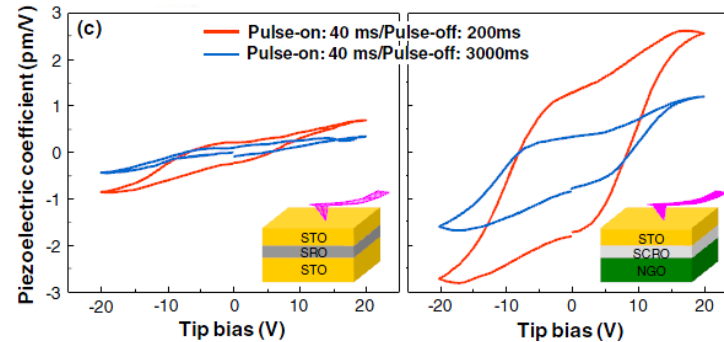
"New" ferroelectrics

Ultra-thin BTO



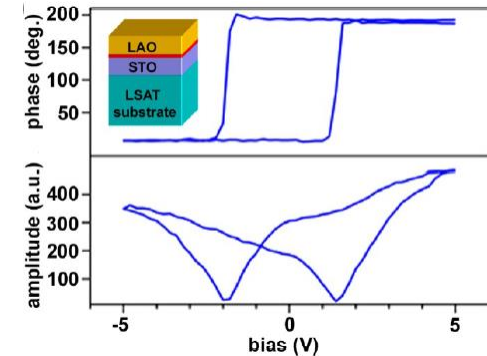
Gruverman et al., Nano Lett. 9, 3539 (2009)

Strained and strain-free STO



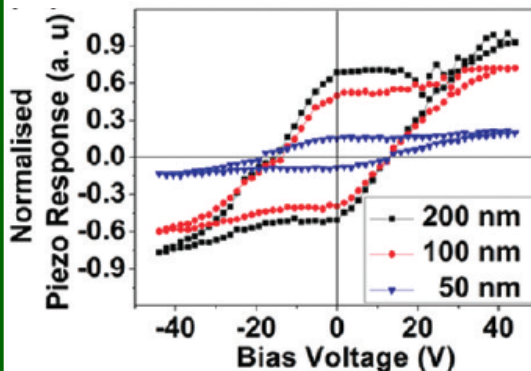
Jang et al., PRL 104, 197601 (2010)

LAO/STO



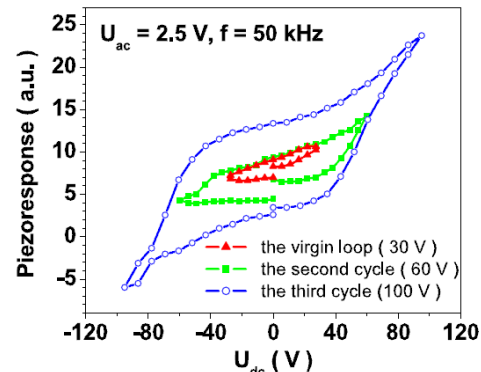
Bark et al., Nano Lett. 12, 1765 (2012)

Sb₂S₃



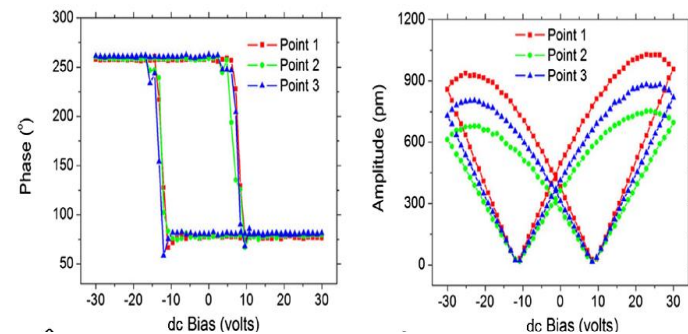
Varghese et al., Nano Lett. 12, 868 (2012).

ZnO



Bidkin et al., Nanotechnol. 21, 235703 (2010)

Aortic walls



Liu et al., PRL 108, 078103 (2012).

$$V = V_{dc} + V_{ac} \sin(\omega t)$$

d_{eff} : effective piezoelectric constant

k : stiffness

C' : Capacitance gradient

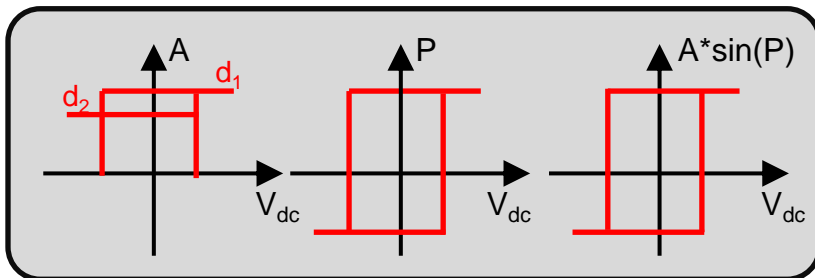
Piezoelectricity

$$x = d_{eff} V$$

$$x = d_{eff} [V_{dc} + V_{ac} \sin(\omega t)]$$

$$x_{\omega} = d_{eff} V_{ac} \sin(\omega t)$$

$$x_{\omega} = d_{eff} (V_{dc}) V_{ac} \sin(\omega t) \Rightarrow \text{sample dependent}$$



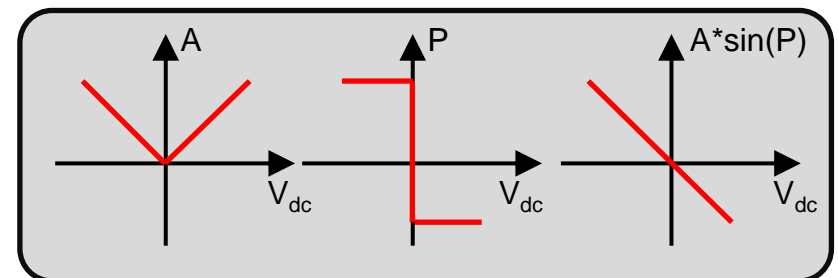
Electrostatics

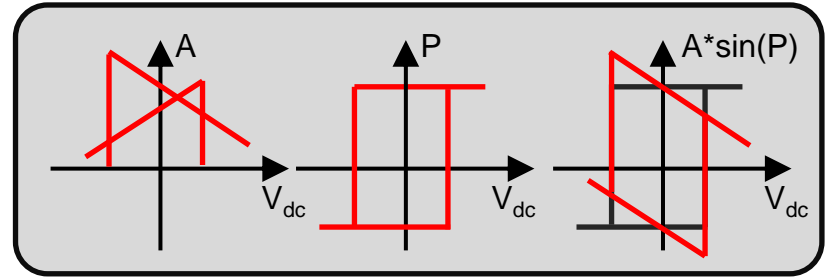
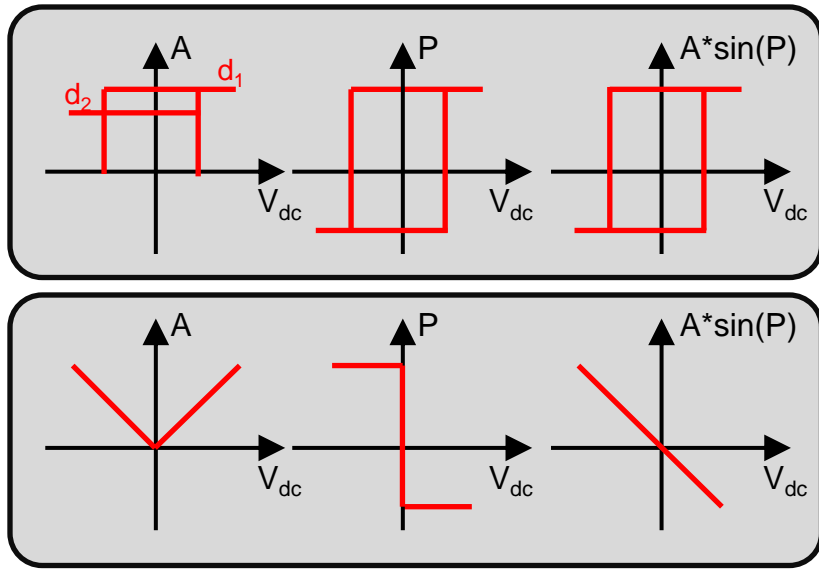
$$x = \frac{1}{2} k^{-1} C' V^2$$

$$x = \frac{1}{2} k^{-1} C' [V_{dc} + V_{ac} \sin(\omega t)]^2$$

$$x_{\omega} = k^{-1} C' V_{dc} V_{ac} \sin(\omega t) \Rightarrow \text{tip and sample dependent}$$

$$x_{2\omega} = \frac{1}{4} k^{-1} C' V_{ac}^2 \cos(2\omega t)$$





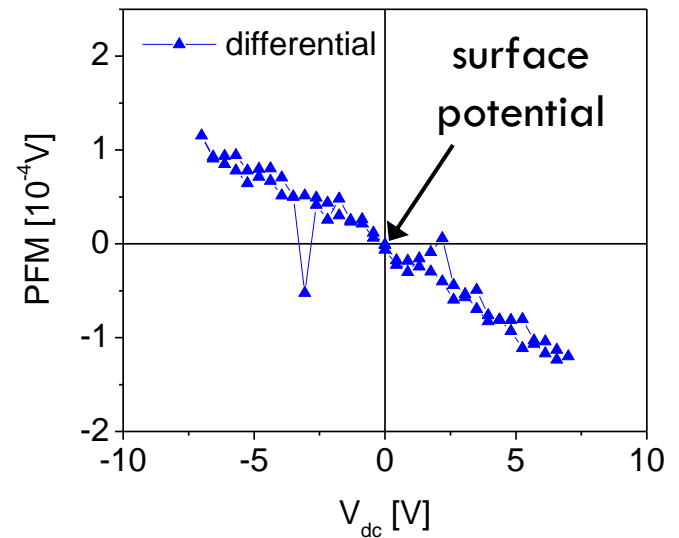
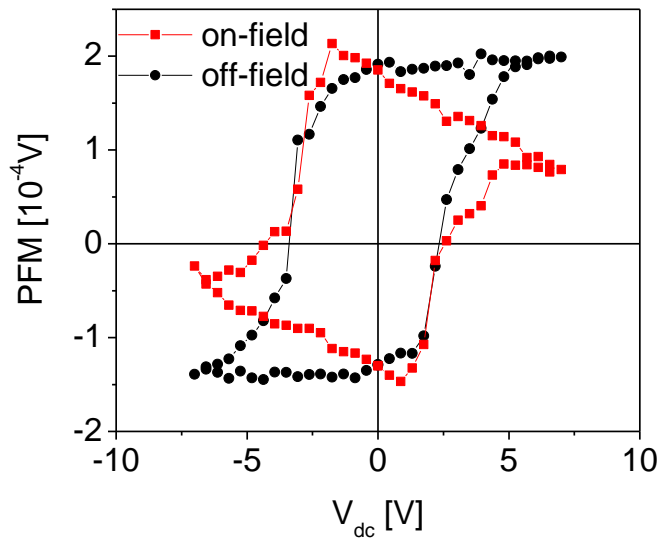
PZT thin film

On-field: $V_{dc} + V_{ac}$

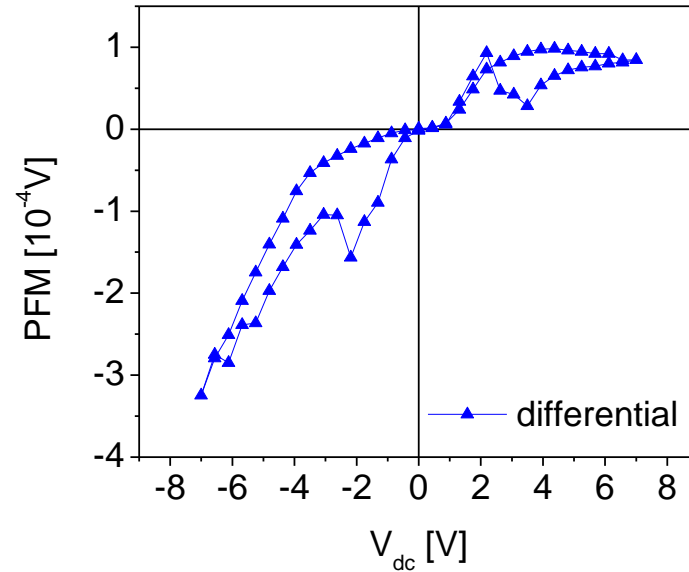
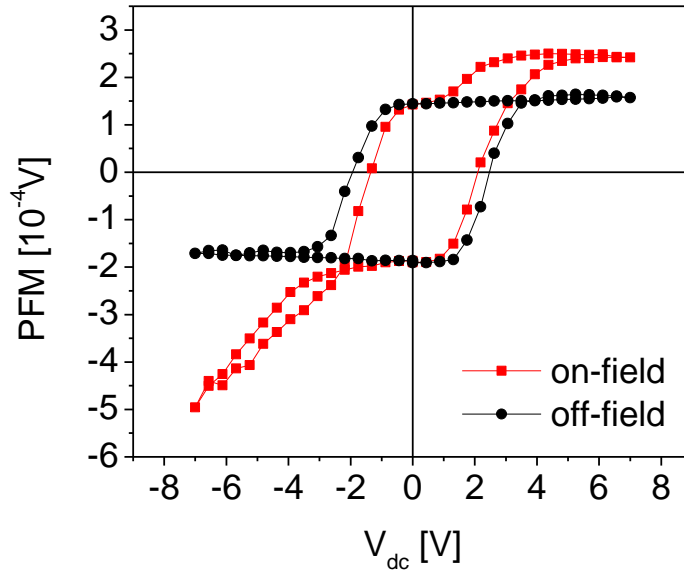
- Piezoresponse
- Electrostatics

Off-field: V_{ac} only

- Piezoresponse

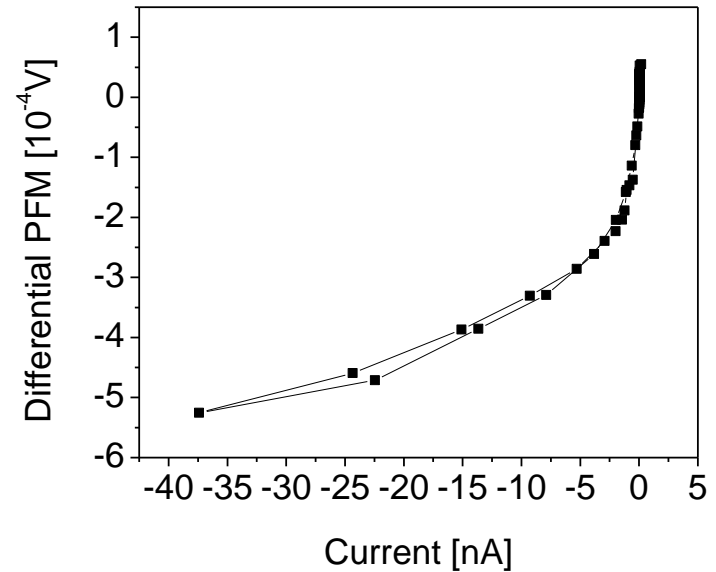
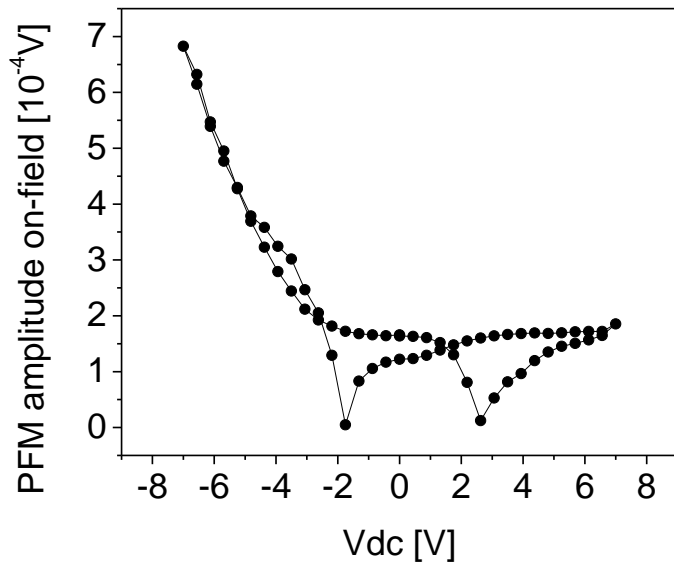
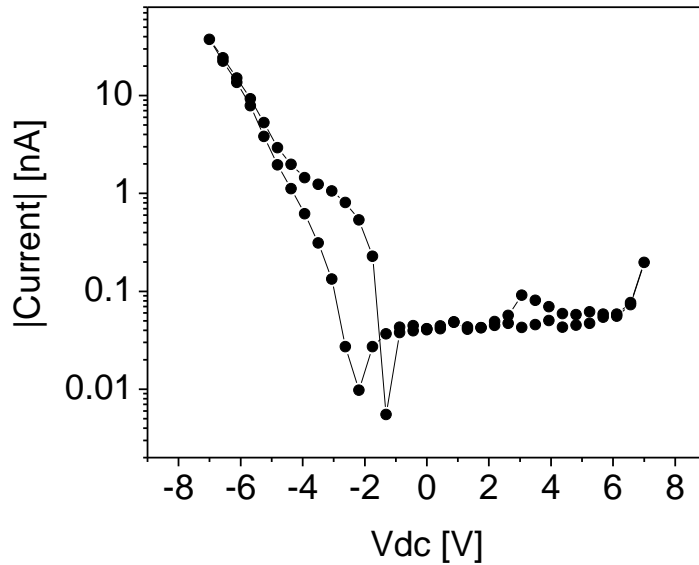


BFO thin film



- Loop shape deviates strongly from the ideal case
- BFO thin films have often problems with leakage (high current flow)

BFO thin film



Joule heating

$$x = \beta R I^2$$

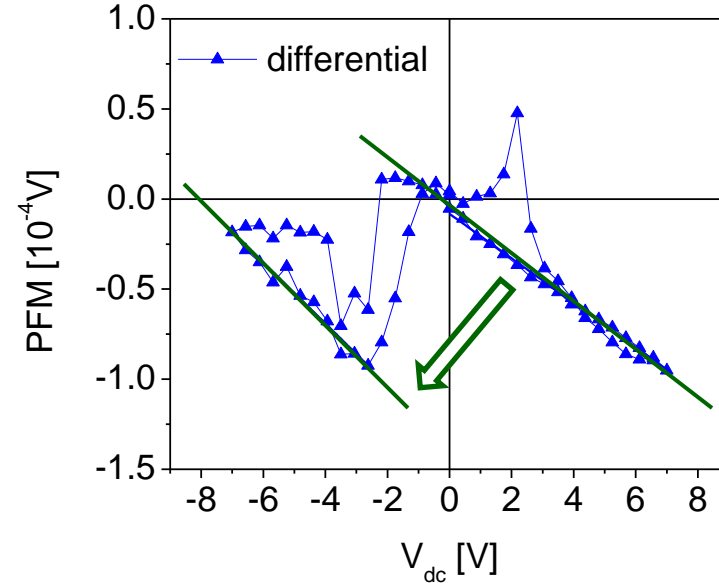
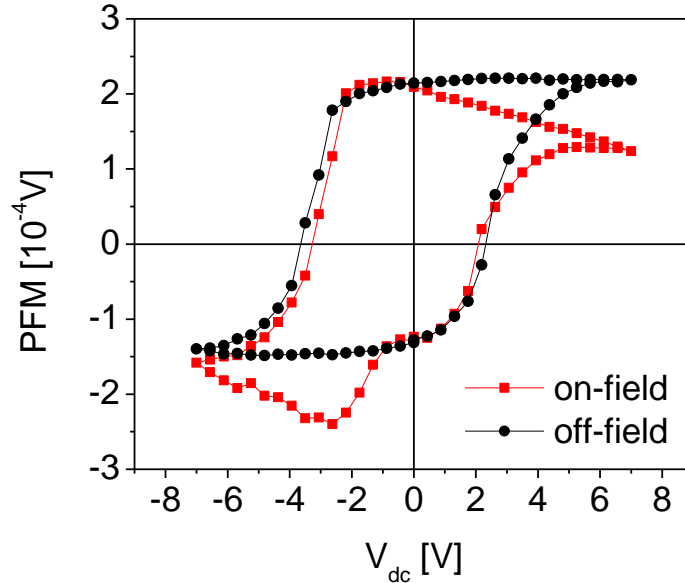
β : Joule
heat
transduction
coefficient

$$x = \beta R [I_{dc} + I_{ac} \sin(\omega t)]^2$$

$$x_{\omega} = 2\beta R I_{dc} I_{ac} \sin(\omega t)$$

$$x_{2\omega} = -\frac{1}{2} \beta R I_{ac}^2 \sin(2\omega t)$$

PZT thin film

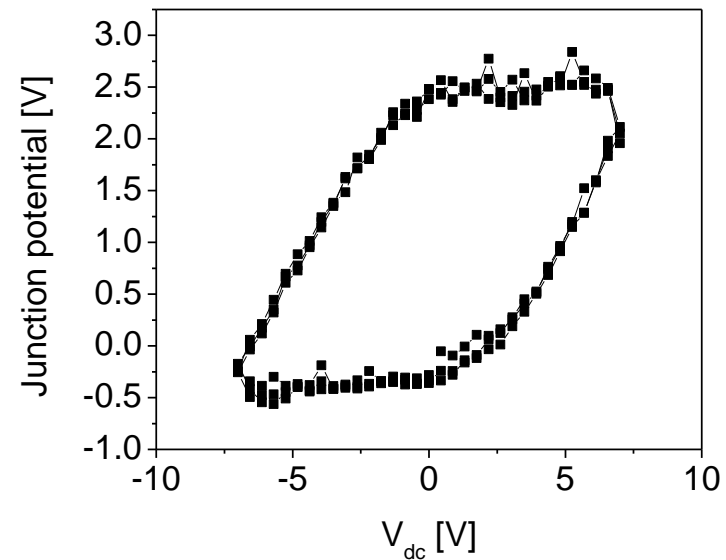
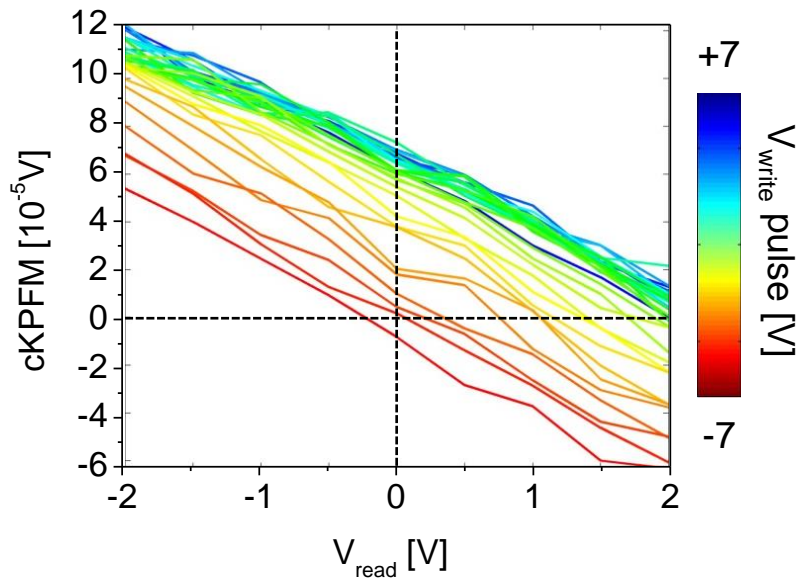


- Loop is shifted down for negative voltages
- Shift in differential loop towards more negative surface potential

How to measure the effect of change in surface potential?

Contact Kelvin Probe Force Microscopy (cKPFM): Change of electrostatic tip-sample interactions after V_{dc} voltage pulses

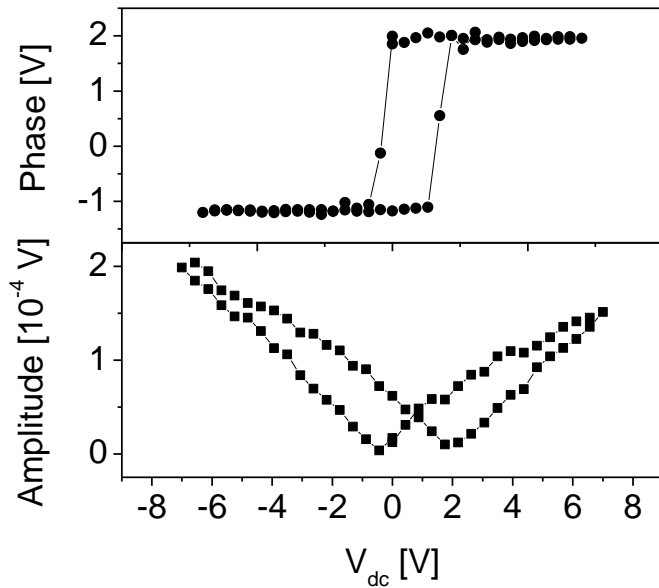
Example: amorphous HfO_2 thin film on Si



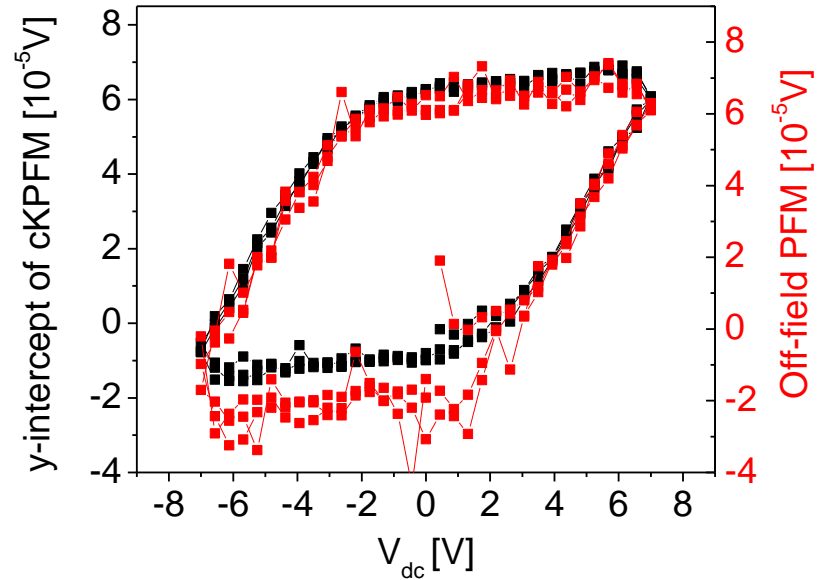
- Hysteretic change in surface potential due to dynamic charge trapping and de-trapping.
- Change in surface potential through charge injection from the tip.

Hysteretic surface charge change + electrostatics = PFM loop!

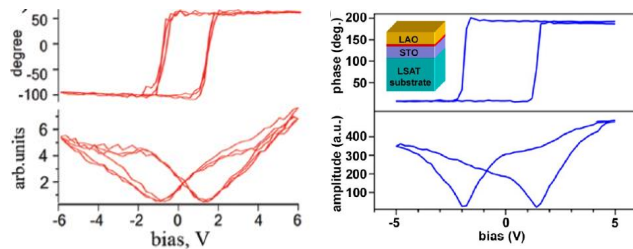
On-field loop



Off-field loop

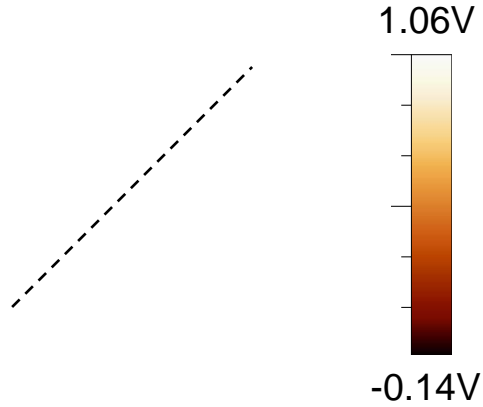


Reminiscent of many PFM loops published on non-traditional ferroelectrics!

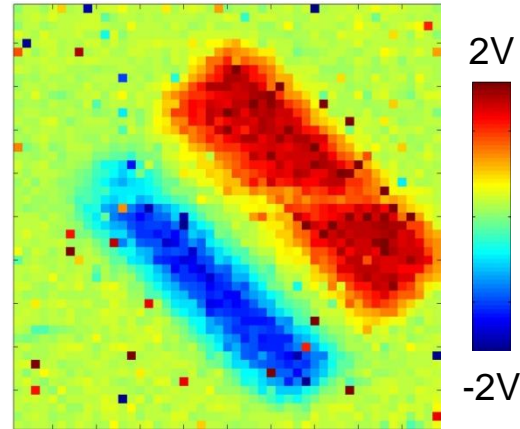


Hysteresis is mainly determined through the change in surface potential.

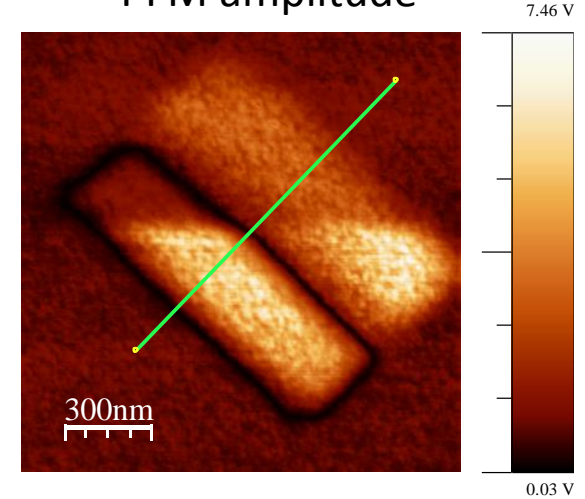
Surface potential via KPFM



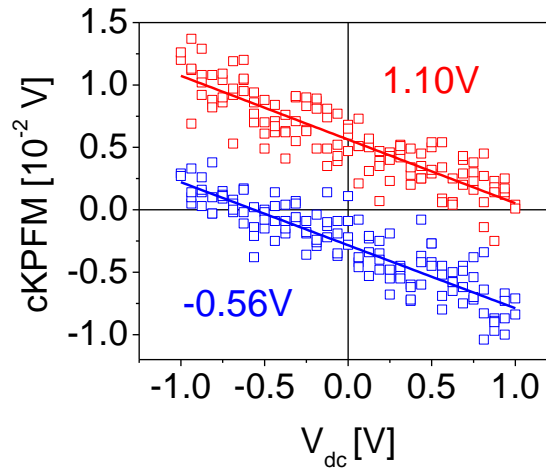
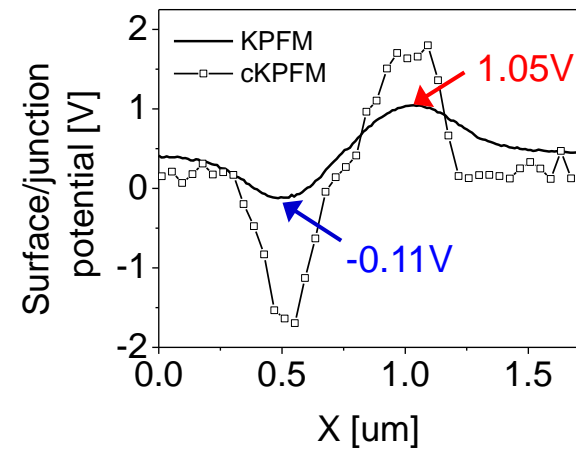
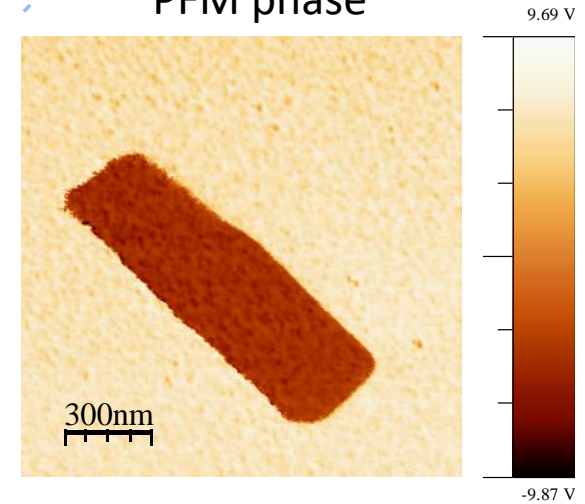
Surface potential via cKPFM



PFM amplitude



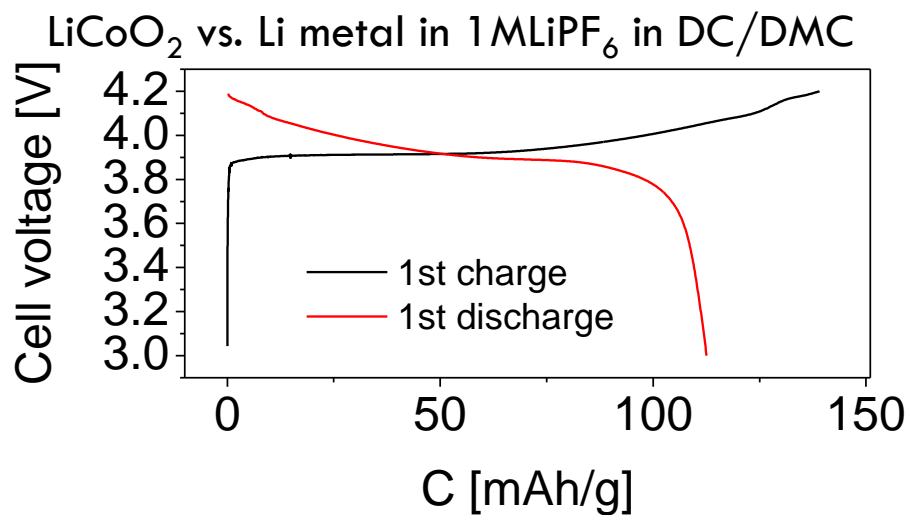
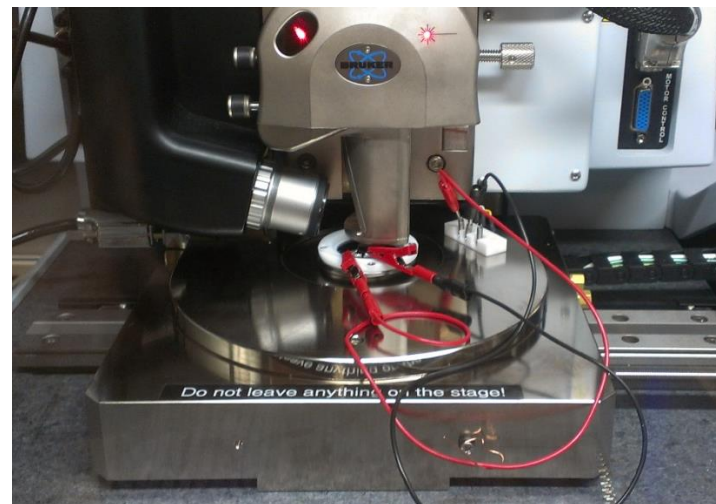
PFM phase



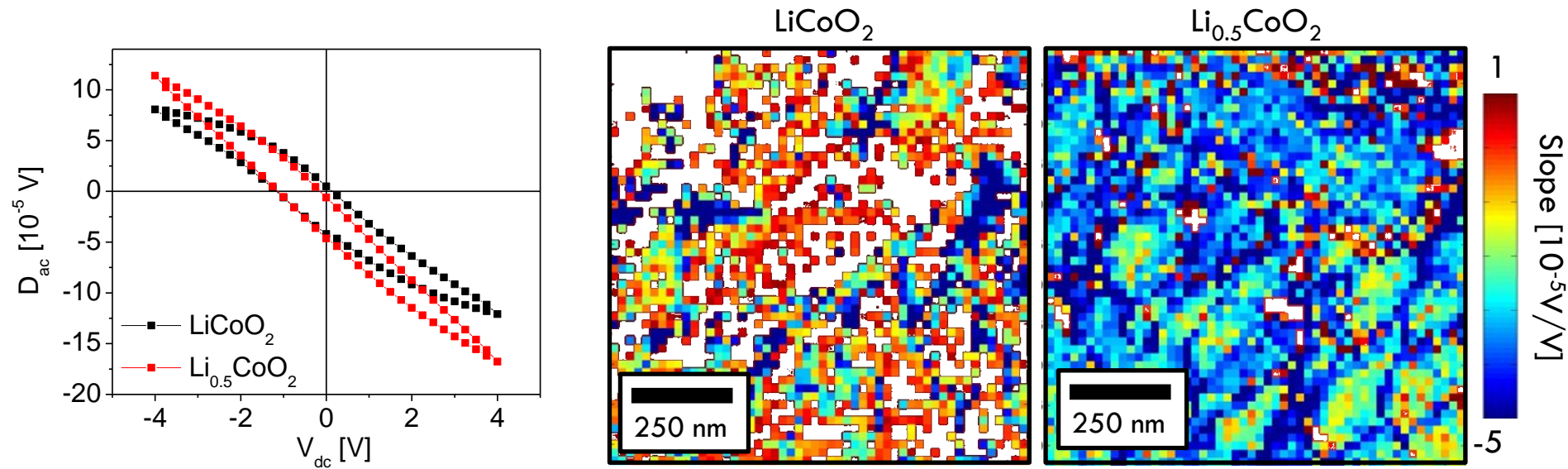
Ferroelectrics

- PFM can be used to image and manipulate domains.
- Piezoelectricity is not the only signal origin during PFM.
- PFM on non-traditional ferroelectrics and ferroelectrics with weak polarization needs to be carefully studied to avoid misinterpretation of data.
- contact KPFM can help to study tip-induced changes in surface potentials and differentiate it from piezoelectric response.

rf-sputtered thin films on $\text{Au}/\text{Al}_2\text{O}_3$



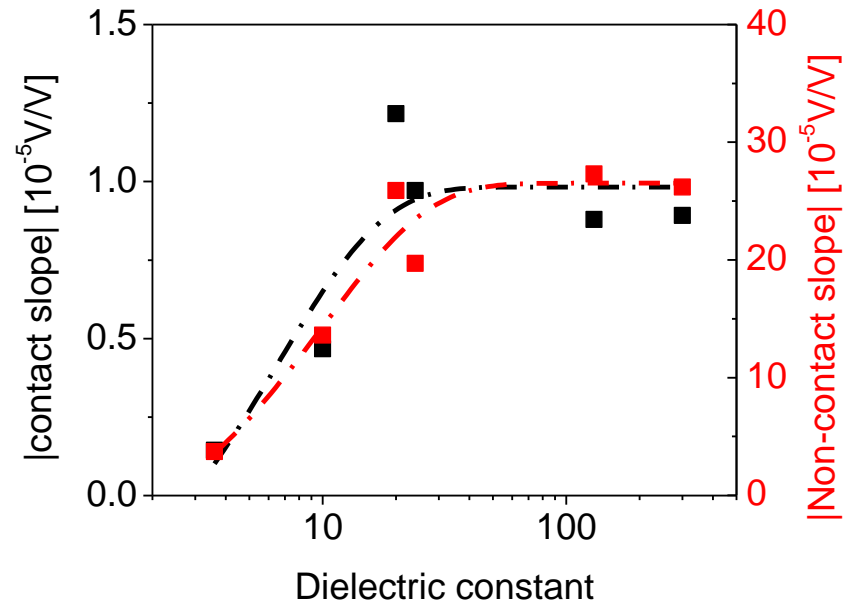
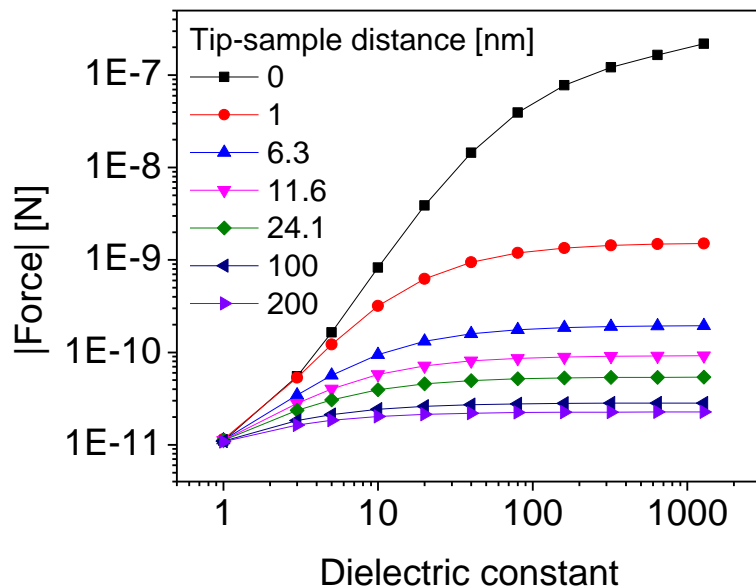
Comparison of LCO in different charging states (ex-situ)



- higher Li-ion mobility when Li-ions are removed from LCO
- Higher slope at the grain boundaries for $\text{Li}_{0.5}\text{CoO}_2$

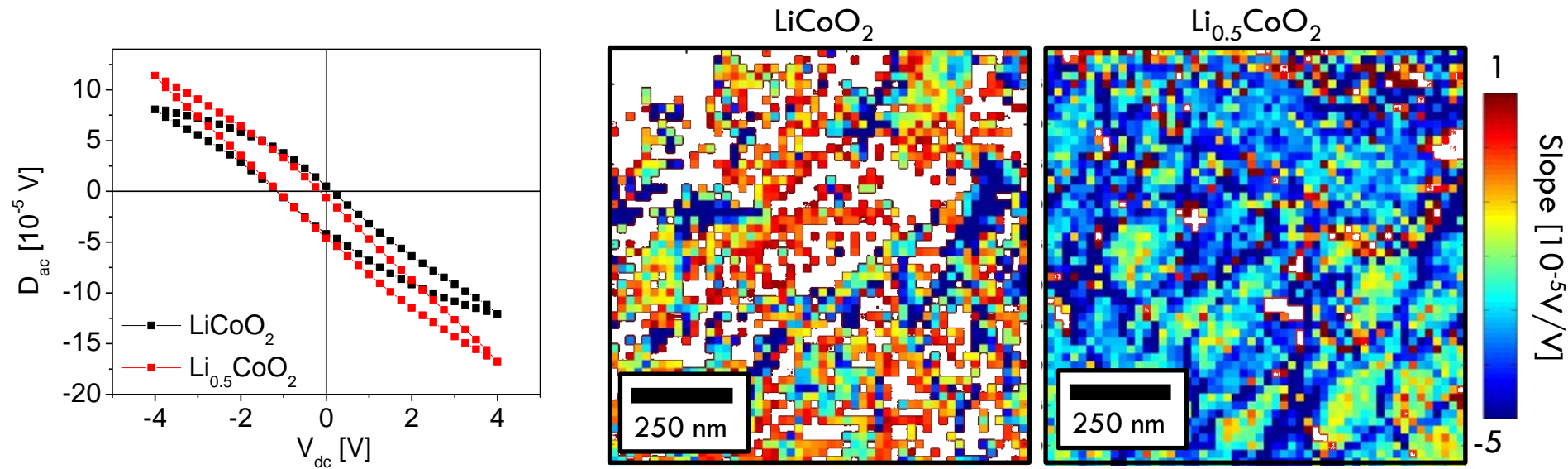
Purely electrostatic case: $x_\omega = k^{-1} C' V_{dc} V_{ac} \sin(\omega t)$

Slope is proportional to ϵ in non-contact and contact



- Ceramics/substrates with known dielectric constant
- Non-contact and contact can be treated equivalent
- Slope is proportional to ϵ if $\epsilon < 20$

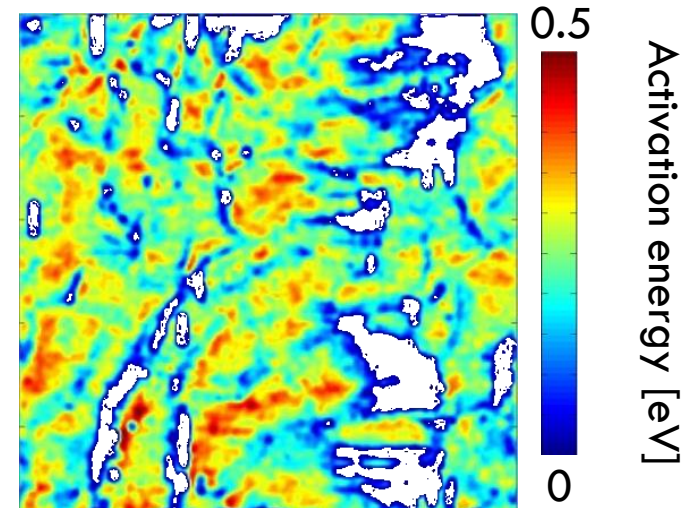
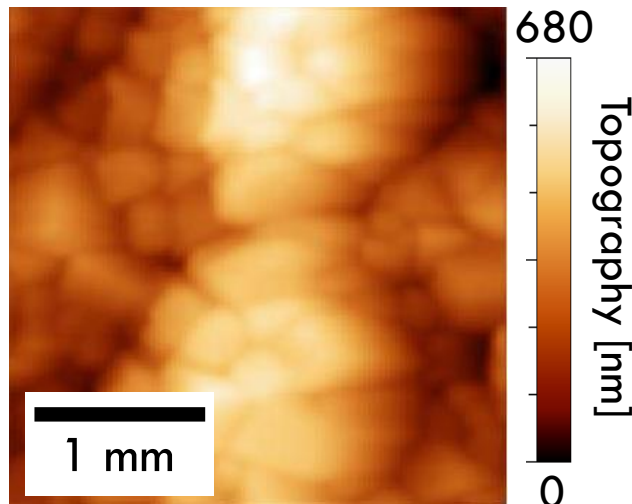
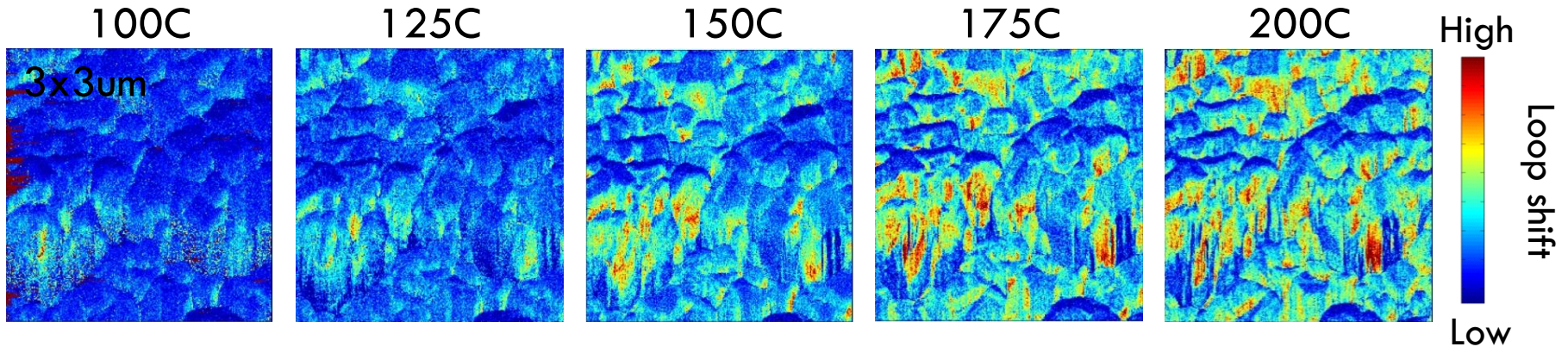
Comparison of LCO in different charging states (ex-situ)



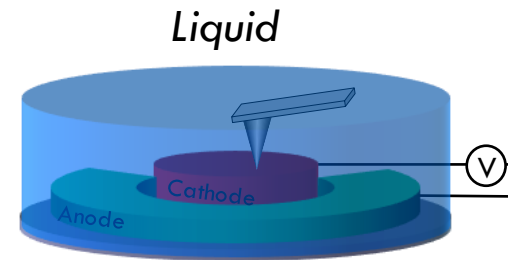
- higher Li-ion mobility when Li-ions are removed from LCO
- Higher ϵ at the grain boundaries for $\text{Li}_{0.5}\text{CoO}_2$

Suggesting that the Li-ion removal happens different in grains and grain boundaries

Temperature dependent measurement of γ -intercept at $0V_{dc}$

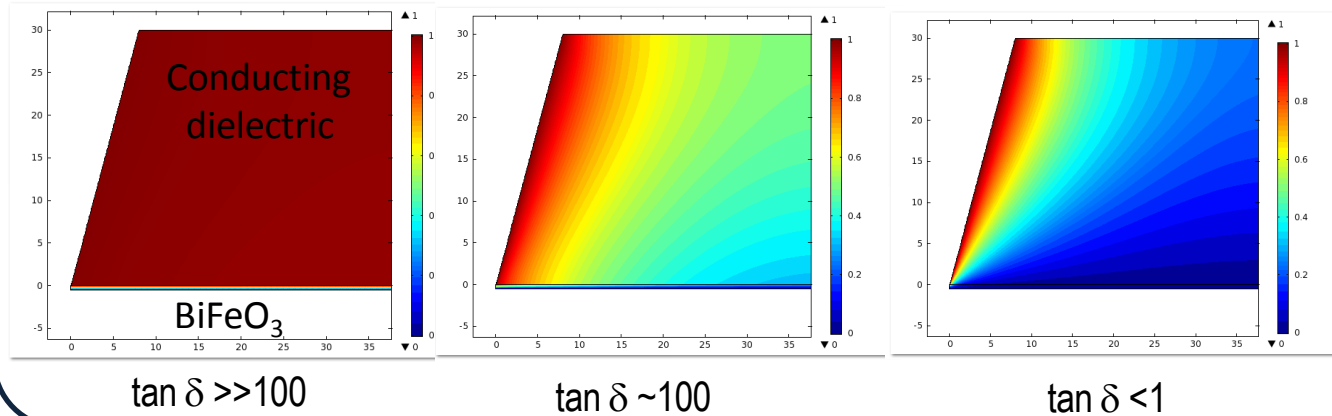


Ultimate dream:
Correlation of nano- and macroscale
properties through in-situ experiments



Liquid environment challenging for SPM → different talk

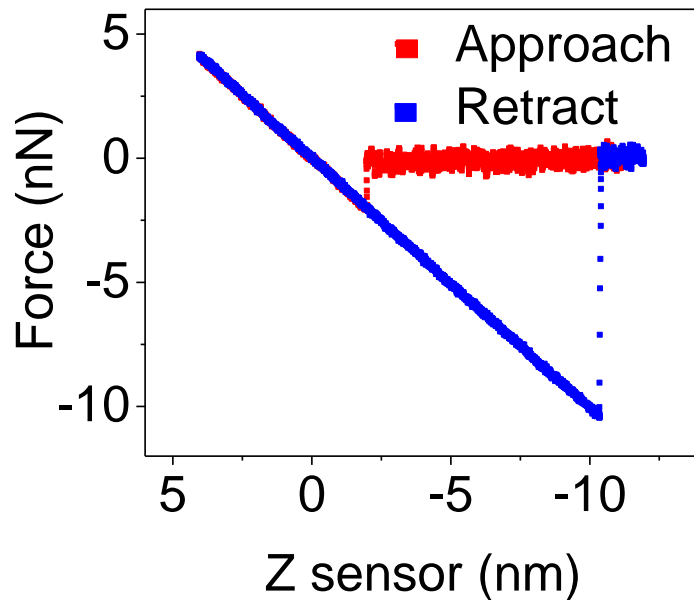
Electric potential



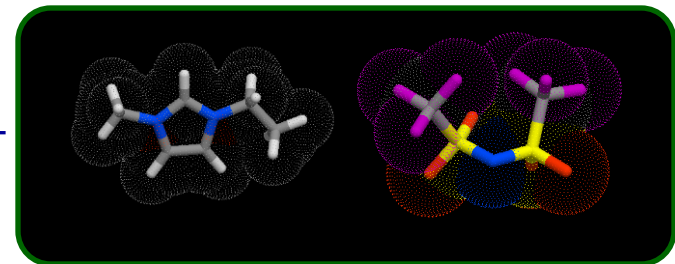
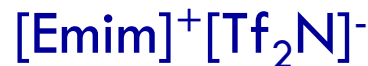
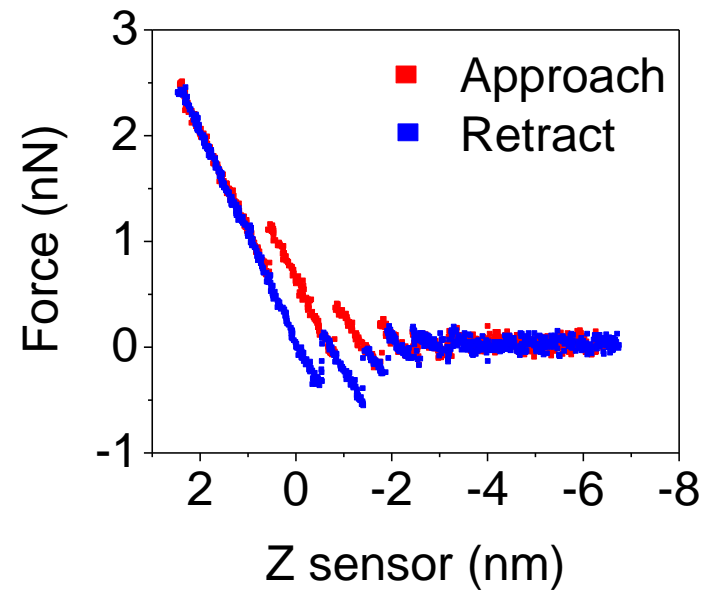
$$\tan \delta = \frac{\omega \epsilon'' + \sigma}{\omega \epsilon'}$$

AFM force spectroscopy to study the electric double layer formed by room temperature ionic liquids on HOPG

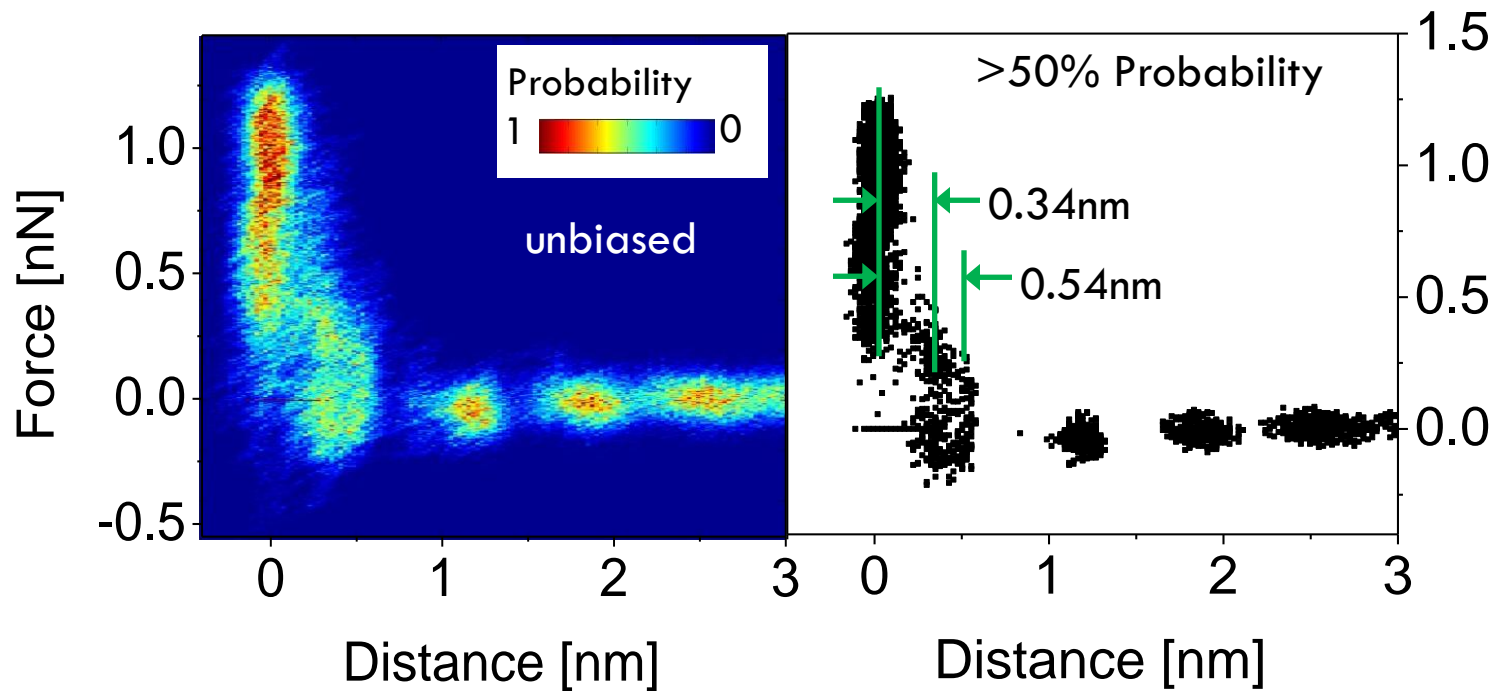
In air



In RTIL

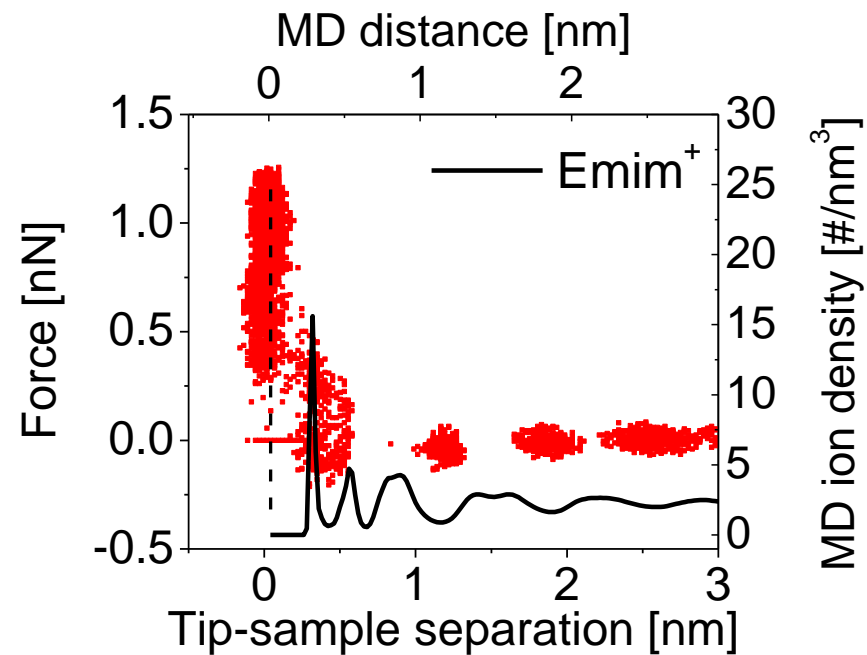
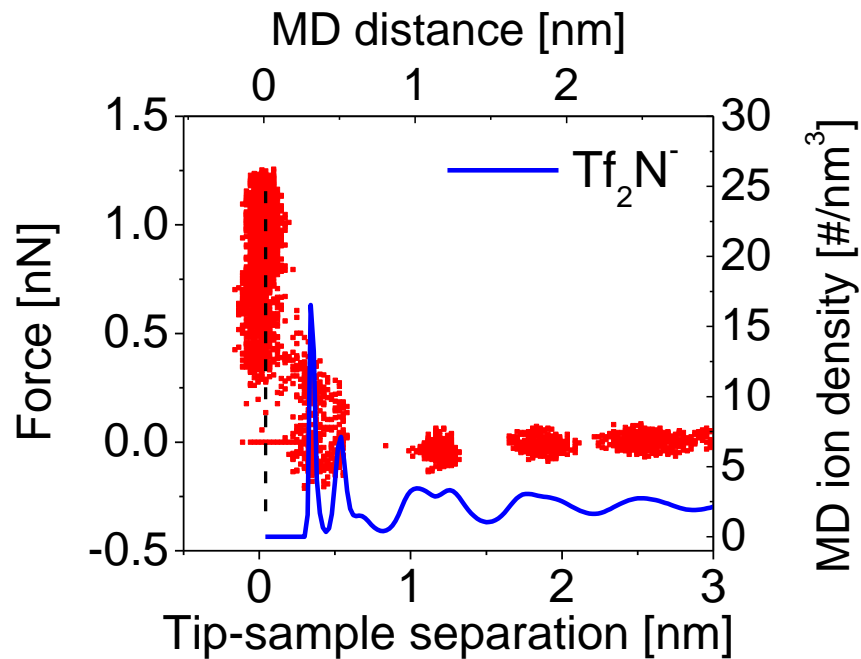


Force spectroscopy on model surfaces (HOPG) in Emim-Tf₂N

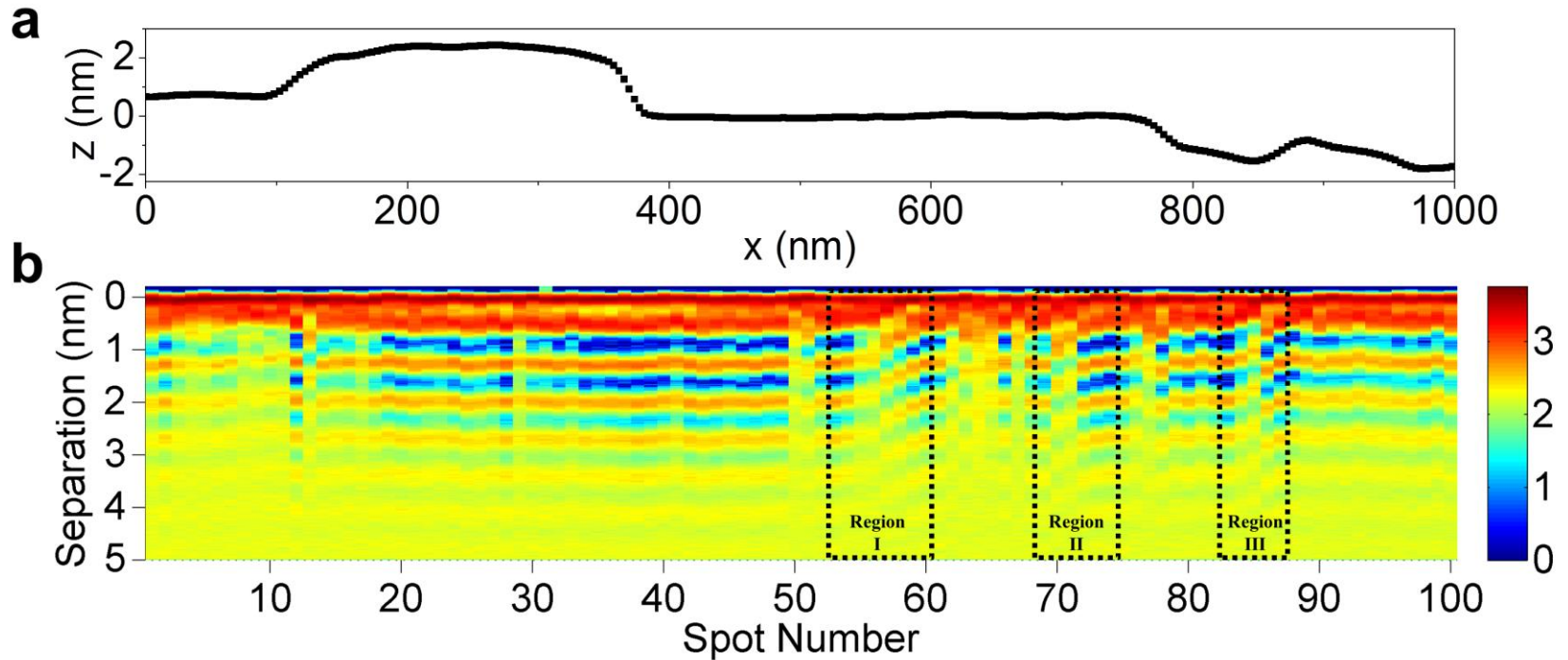


Statistical approach to identify ion positions.

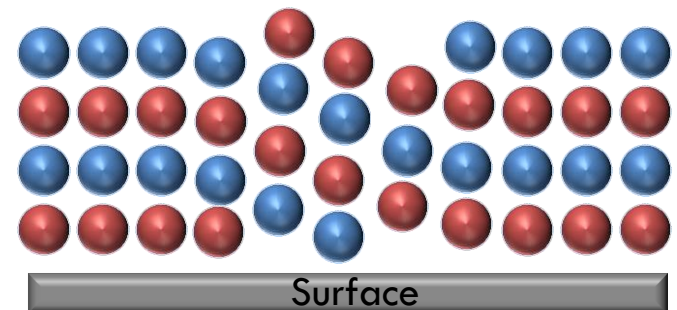
Comparison of F-d with MD:

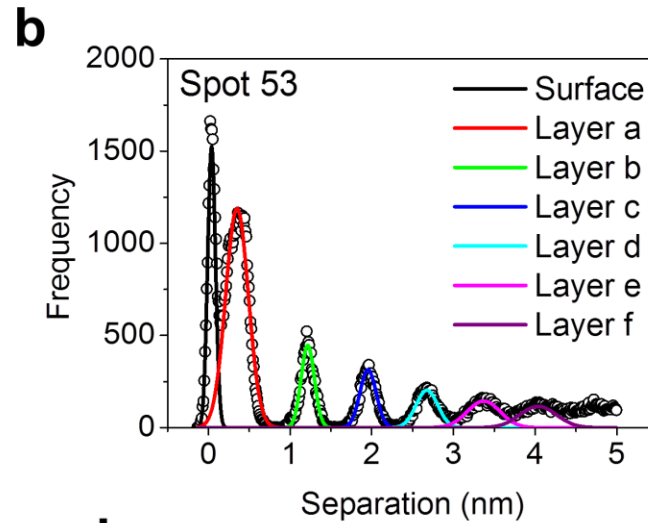
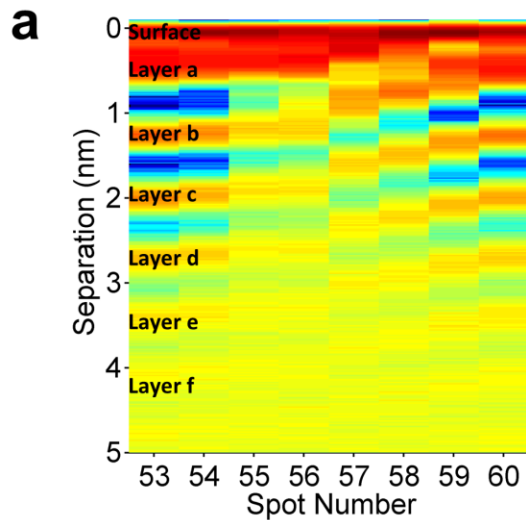


The measured ion positions coincide with anion positions.

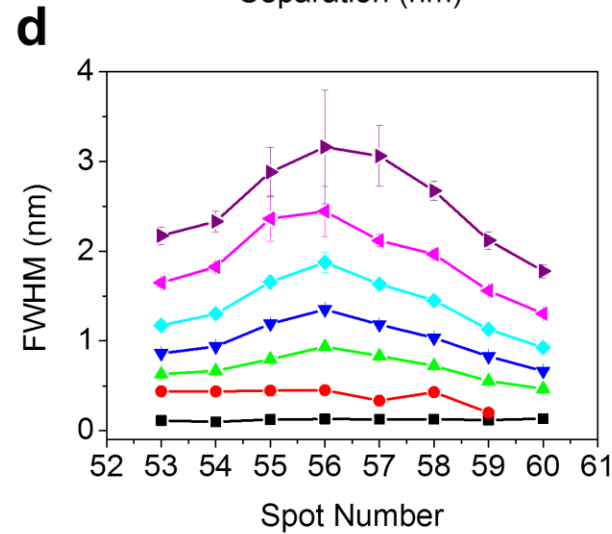
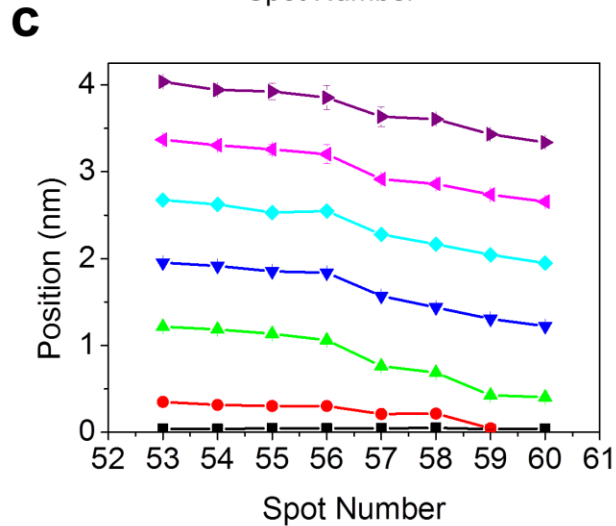


Edge dislocations observed in the ionic layering in regions with no step edges.





Width of ion
layer as
measure for
disorder.



Disorder is
increased when
ions are not in
their equilibrium
position.

Energy storage systems

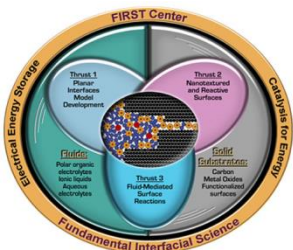
- SPM can be used to study energy storage systems.
- Changes in dielectric properties can help looking at ionic transport.
- Force spectroscopy can “see” ion layering in the EDL for ionic liquids.
- Ionic liquids behave as liquid crystals and show ordering defects.
- Functionality of extended defects in future studies.



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