

Point defects and surfaces of transparent electrodes studied by DFT-calculations

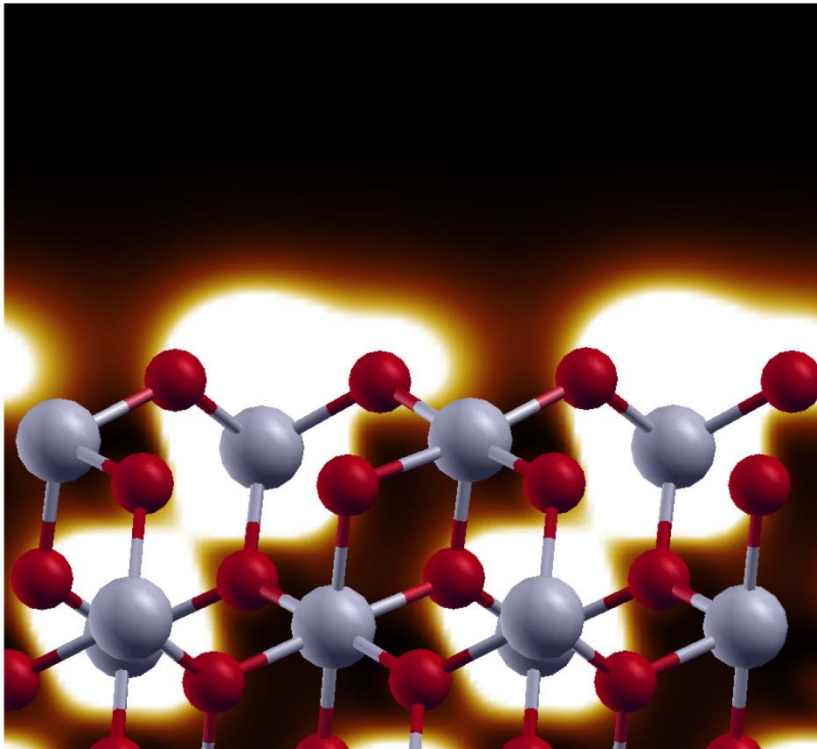


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FACHGEBIET

MATERIAL
MODELLIERUNG



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Andreas Klein², Manuel Diehm¹,
Arno Fey¹, **Karsten Albe¹***

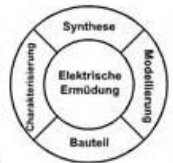
TU Darmstadt

¹FG Materialmodellierung

²FG Oberflächenforschung

FB Material- und Geowissenschaften

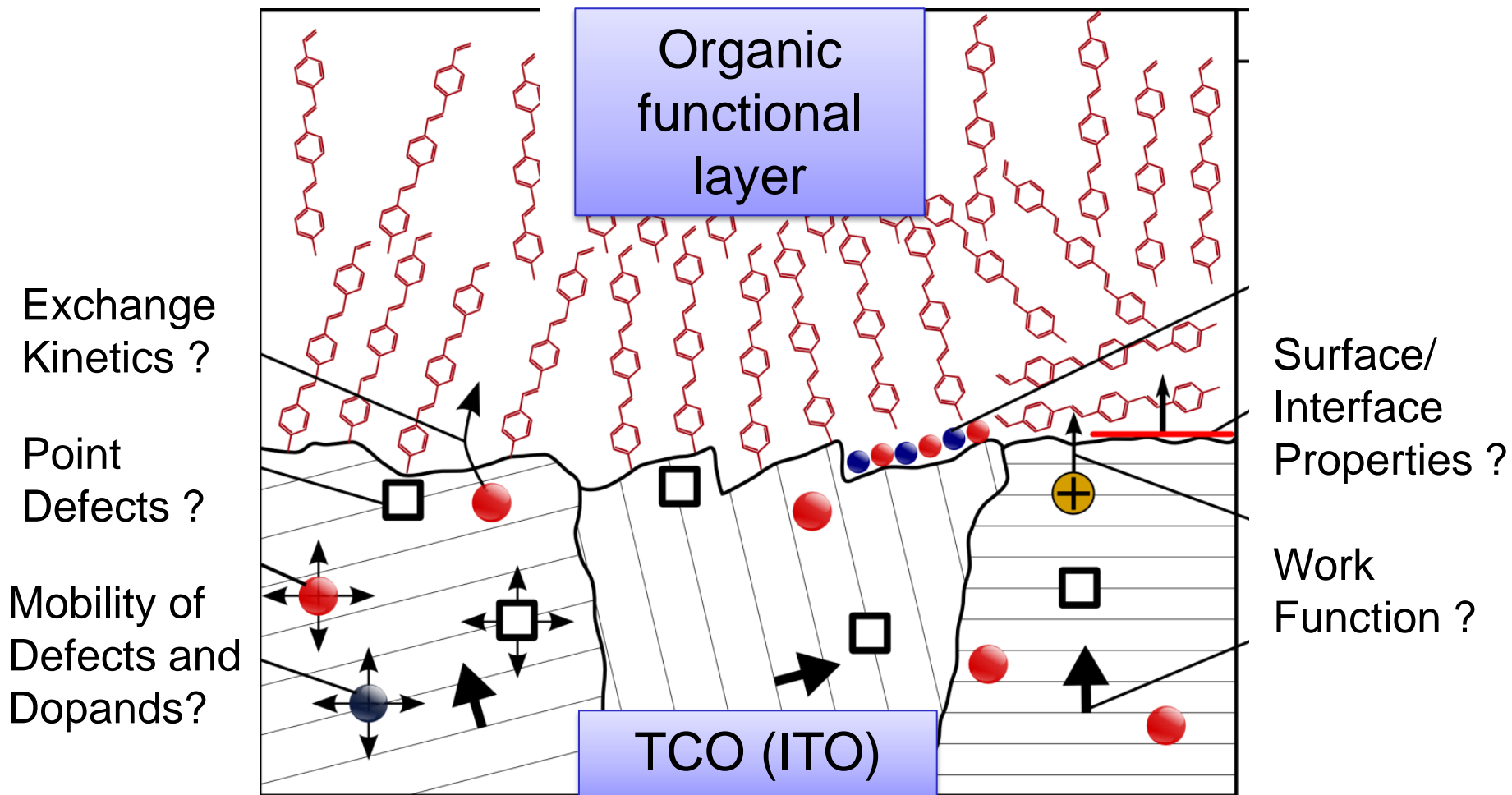
SFB 595



Deutsche
Forschungsgemeinschaft

DFG

OLEDs and TCOs: Open Questions

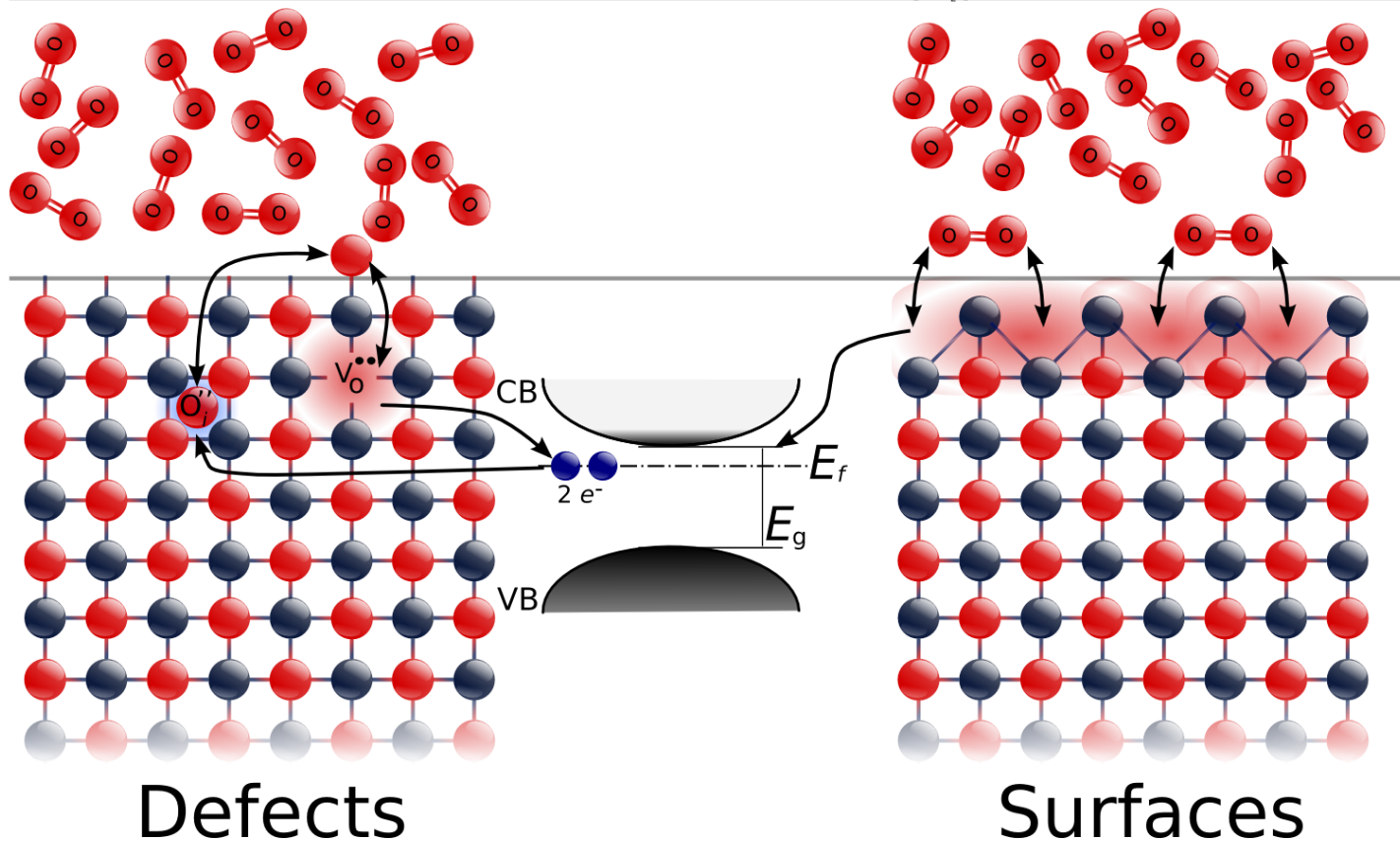


Defect Thermodynamics



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$$\mu^{\circ}(p_{O_2}, T) = \frac{1}{2} \mu_{O_2}^{\circ}(p_{O_2}^{\circ}, T) + \frac{1}{2} RT \ln \left(\frac{p_{O_2}}{p_{O_2}^{\circ}} \right)$$



$$\gamma = (E_{slab} - TS_{slab} - \sum \mu_i n_i) / 2A$$

Surface Stresses and Surface Energies



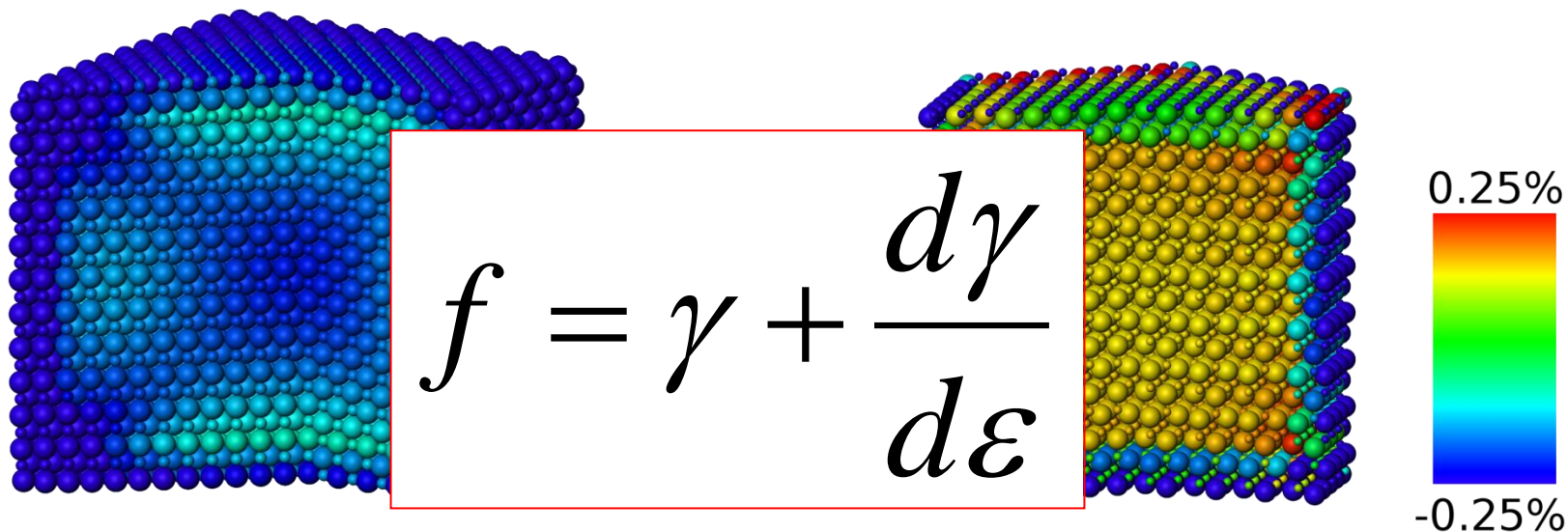
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$$G = G_0 + \gamma A$$

$$\begin{aligned} dG &= d(\gamma A) = \gamma dA + A d\gamma \\ &= \left(\gamma + A \left(\frac{\partial \gamma}{\partial A} \right) \right) dA \end{aligned}$$

Role of Surface Stresses

- MgO (compressive): GaN (tensile)

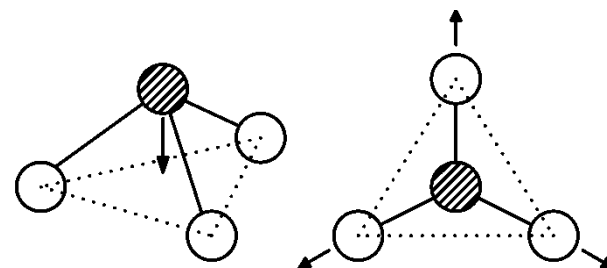


CHEMPHYSICHEM MINIREVIEWS

DOI: 10.1002/cphc.201200257

Size-Dependent Lattice Expansion in Nanoparticles: Reality or Anomaly?

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Calculated Surface Stresses (DFT)

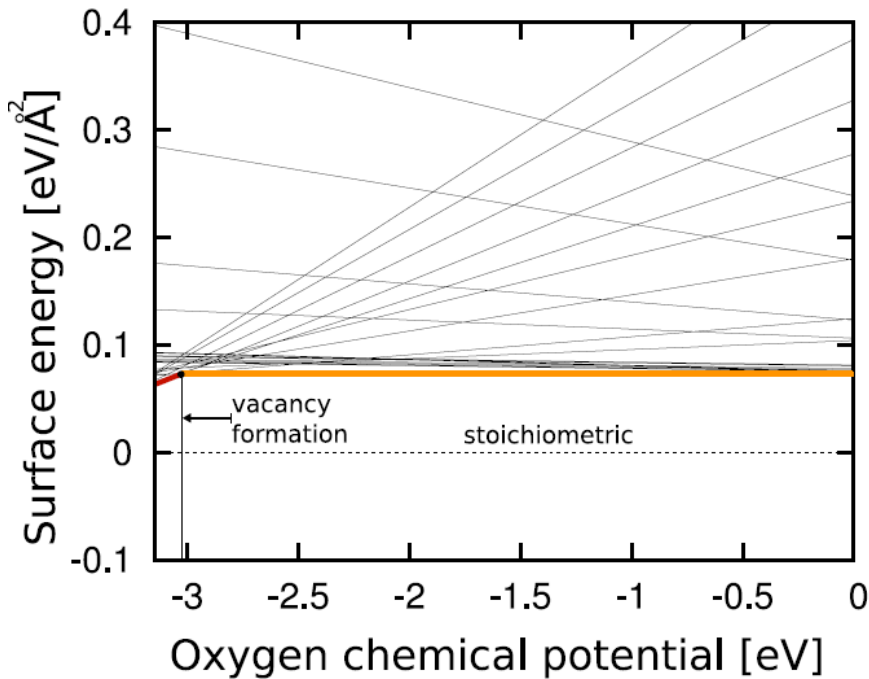


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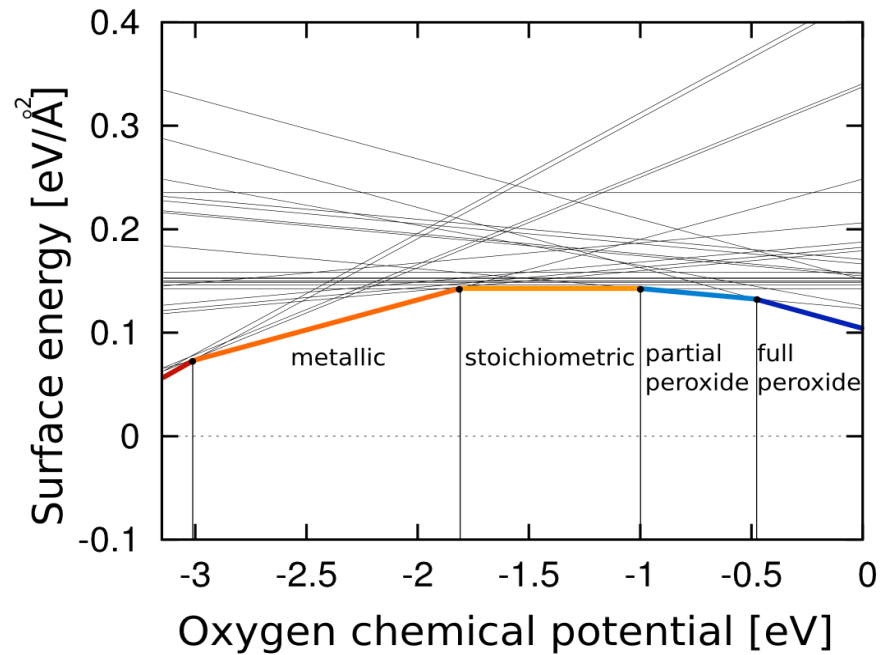
Material	Surface	Condition	x-direction	y-direction	γ	f_x f_y	
						(eV Å ⁻²)	
hex-Si	(0001)		$\langle 1100 \rangle$	$\langle 1210 \rangle$	0.114	-0.045	-0.039
	($1\bar{1}00$)		$\langle 1\bar{2}10 \rangle$	$\langle 0001 \rangle$	0.086	-0.065	-0.032
	($1\bar{2}10$)		$\langle 1\bar{1}00 \rangle$	$\langle 0001 \rangle$	0.097	-0.043	-0.030
hex-C	(0001)		$\langle 1\bar{1}00 \rangle$	$\langle 1\bar{2}10 \rangle$	0.419	0.003	0.029
	($1\bar{1}00$)		$\langle 1\bar{2}10 \rangle$	$\langle 0001 \rangle$	0.184	-0.160	0.056
	($1\bar{2}10$)		$\langle 1\bar{1}00 \rangle$	$\langle 0001 \rangle$	0.227	-0.337	0.020
GaN	(1100)		$\langle 1210 \rangle$	$\langle 0001 \rangle$	0.125	-0.106	-0.055
	($1\bar{2}10$)		$\langle 1\bar{1}00 \rangle$	$\langle 0001 \rangle$	0.133	-0.031	-0.030
ZnO	($1\bar{1}00$)		$\langle 1\bar{2}10 \rangle$	$\langle 0001 \rangle$	0.075	-0.088	-0.059
	($1\bar{2}10$)		$\langle 1\bar{1}00 \rangle$	$\langle 0001 \rangle$	0.079	-0.011	-0.038
BeO	($1\bar{1}00$)		$\langle 1\bar{2}10 \rangle$	$\langle 0001 \rangle$	0.109	-0.089	-0.054
	($1\bar{2}10$)		$\langle 1\bar{1}00 \rangle$	$\langle 0001 \rangle$	0.113	-0.039	-0.044
MgO	(001)		$\langle 100 \rangle$	$\langle 100 \rangle$	0.073	0.223	0.223
CdO	(001)		$\langle 100 \rangle$	$\langle 100 \rangle$	0.051	0.210	0.210
KF	(001)		$\langle 100 \rangle$	$\langle 100 \rangle$	0.016	0.037	0.037
CsF	(001)		$\langle 100 \rangle$	$\langle 100 \rangle$	0.011	0.033	0.033
CsCl	(110)		$\langle 110 \rangle$	$\langle 100 \rangle$	0.014	0.058	-0.002
c-SrTiO ₃	(001)	SrO-termination	$\langle 100 \rangle$	$\langle 100 \rangle$	0.166	0.116	0.116

In₂O₃: (111) vs. (001) surfaces

non-polar (type-II)



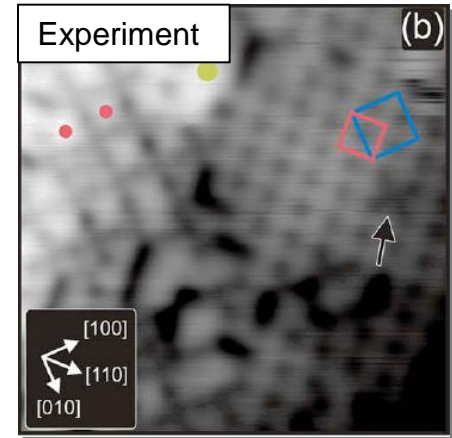
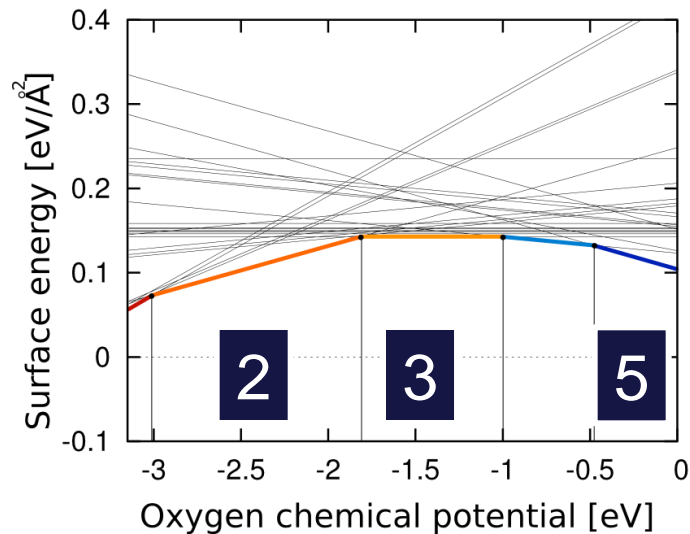
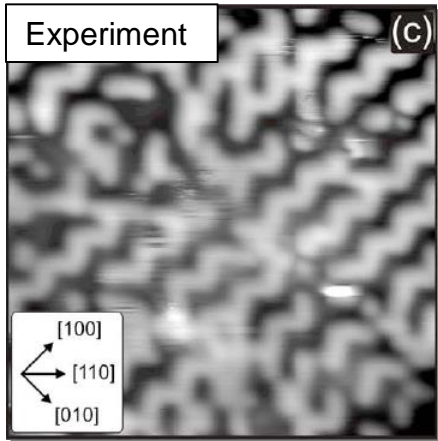
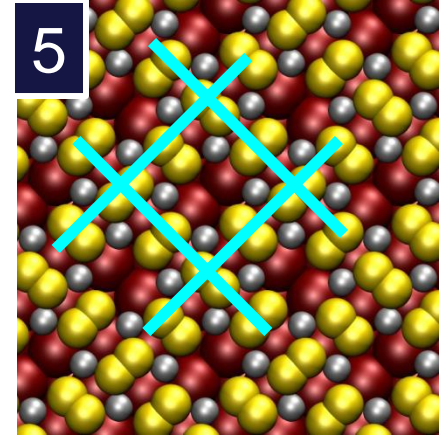
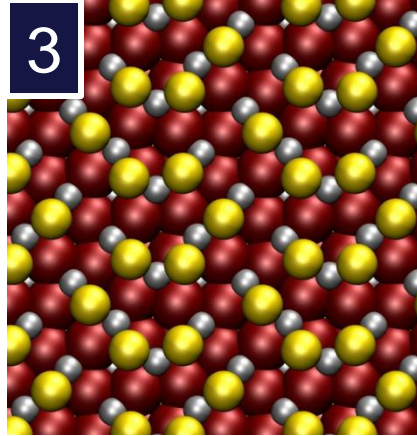
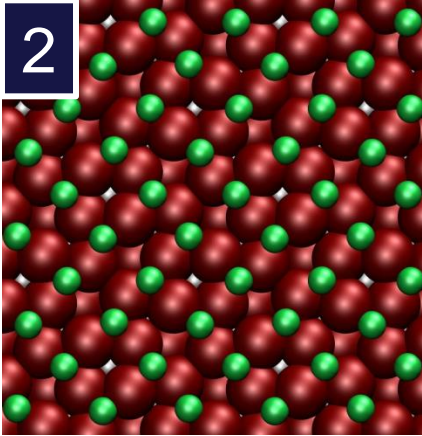
polar (type-IIIa)



- (111) mainly stoichiometric
- Several phase transitions in the case of (001)

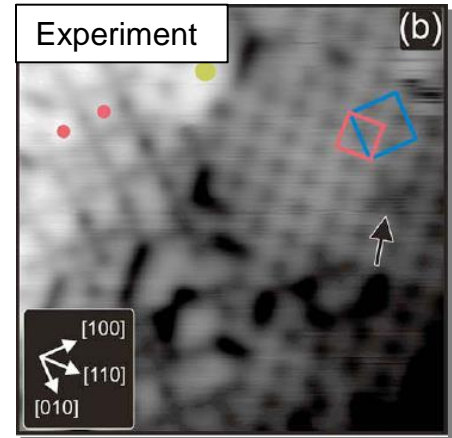
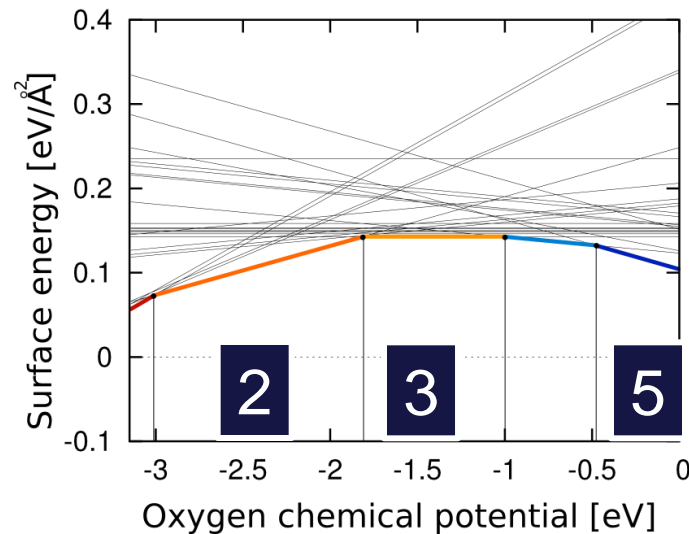
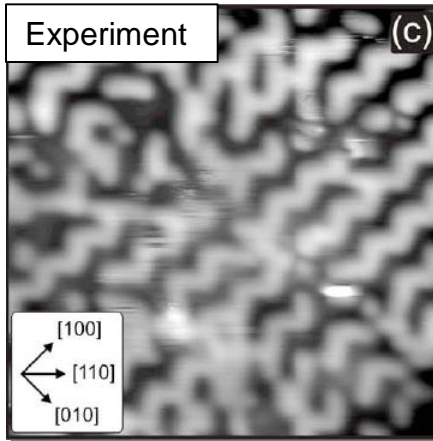
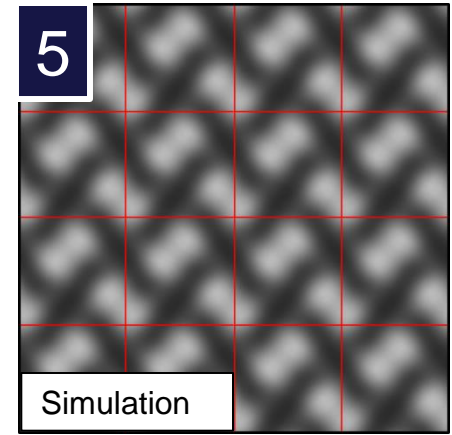
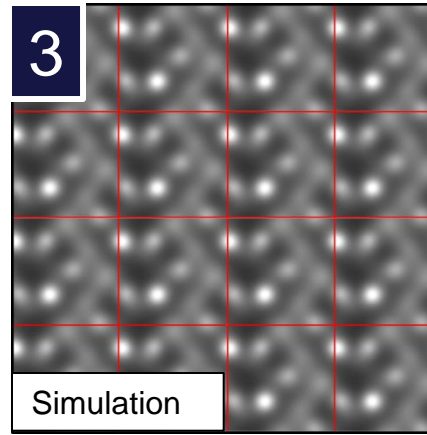
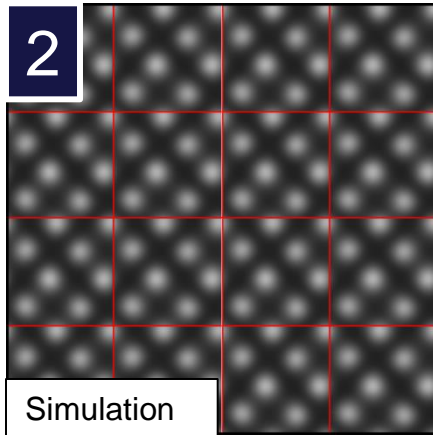
ITO-(001)

Oxygen : ● Indium : ○
Oxygen (1. layer) : ● Indium (1. layer) : ●



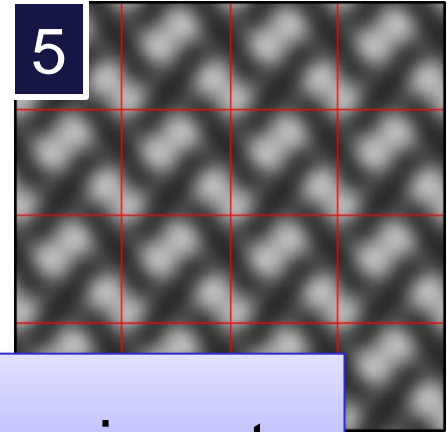
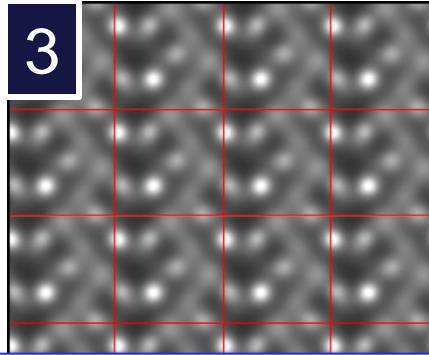
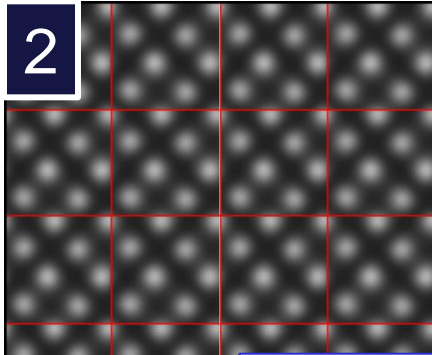
In_2O_3 -(001)

Oxygen : ● Indium : ○
Oxygen (1. layer) : ● Indium (1. layer) : ○



In_2O_3 -(001)

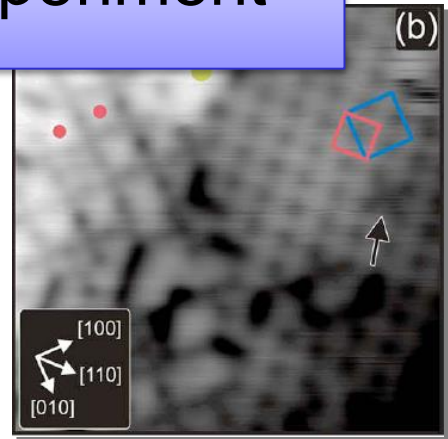
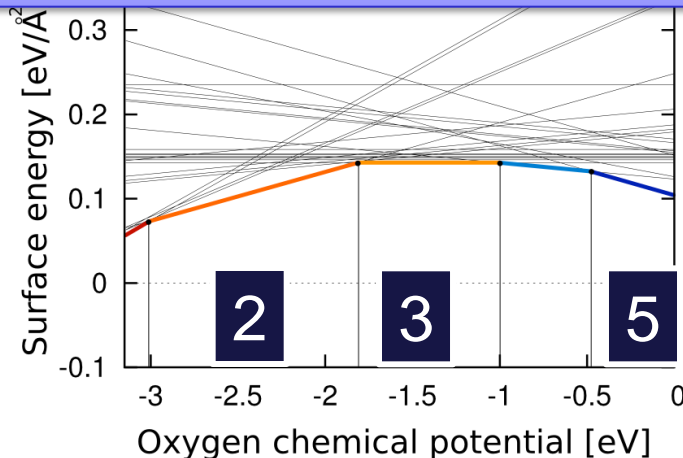
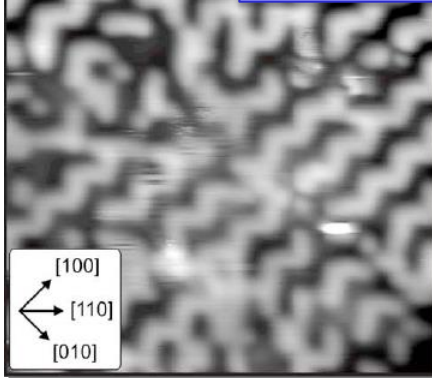
Oxygen : ● Indium : ○
Oxygen (1. layer) : ● Indium (1. layer) : ○



Simulation

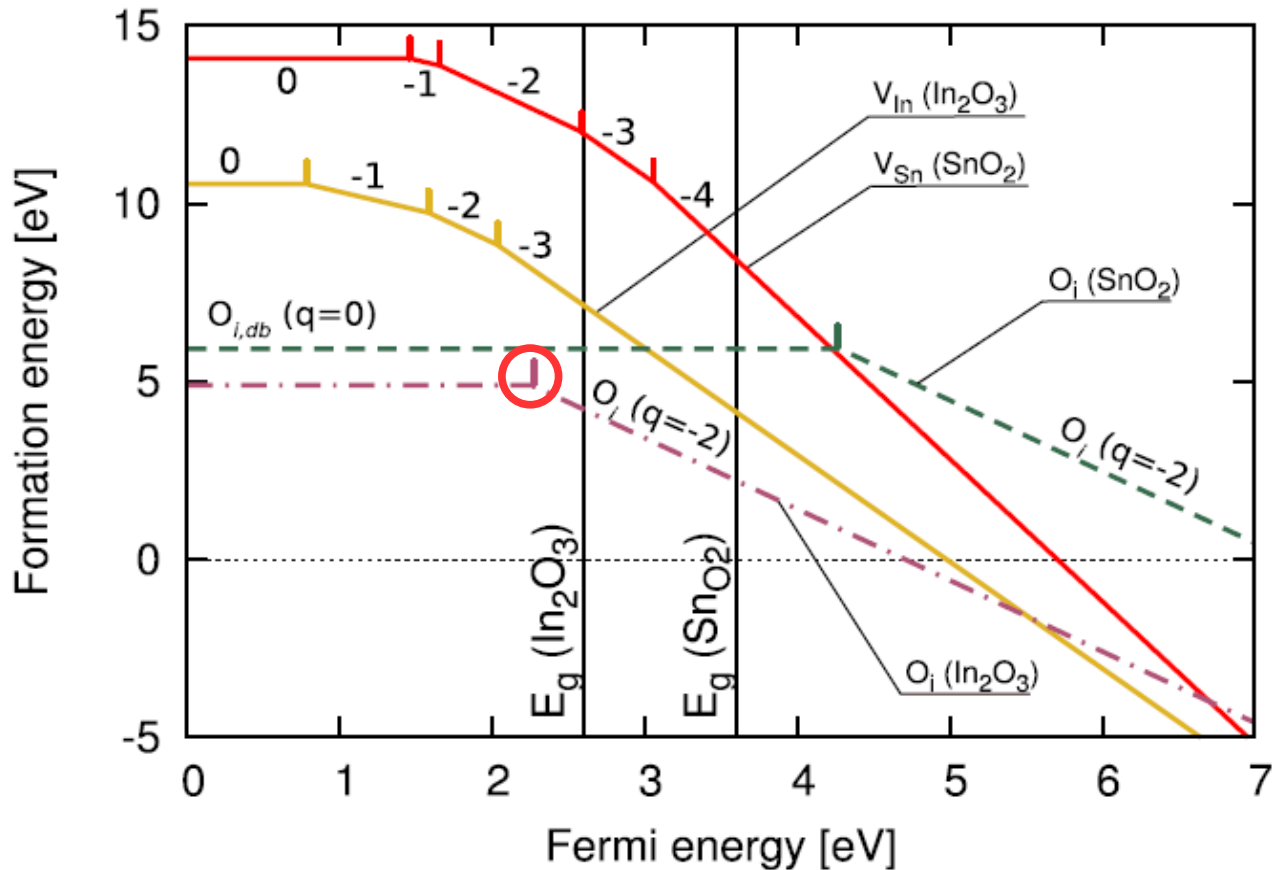
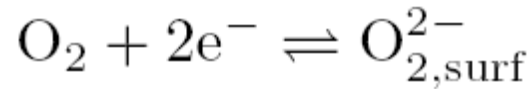
Experiment

➤ No compelling agreement with experiment

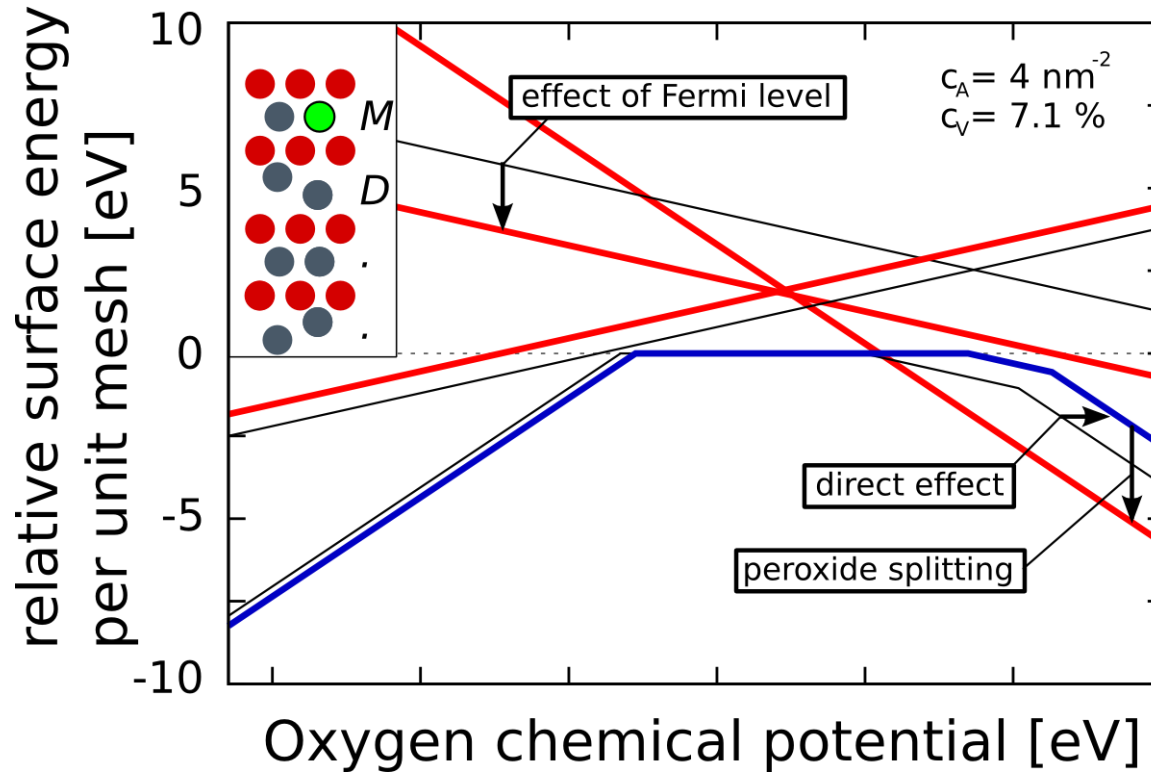




Doping effect at In_2O_3 -(001) ?



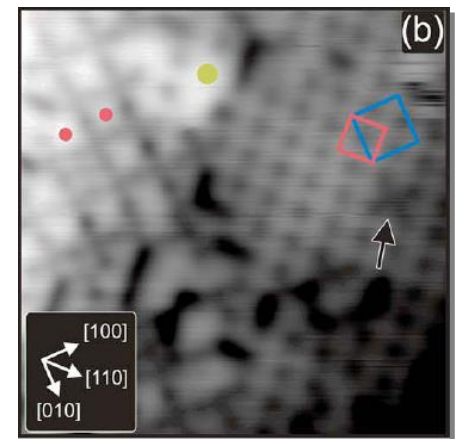
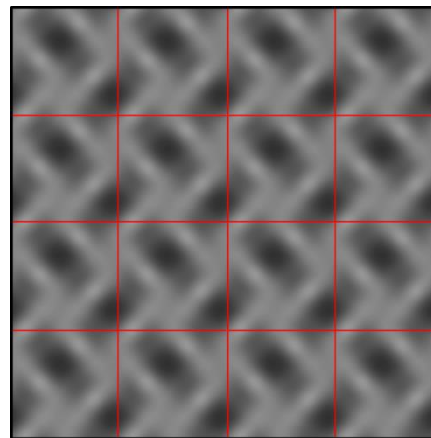
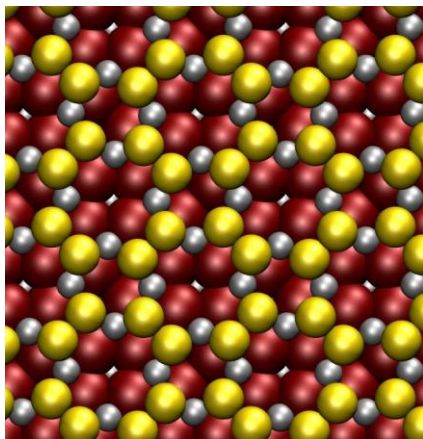
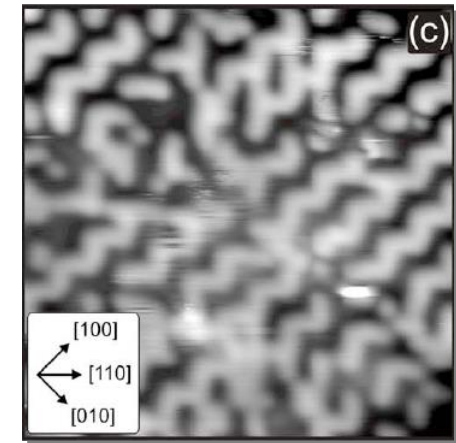
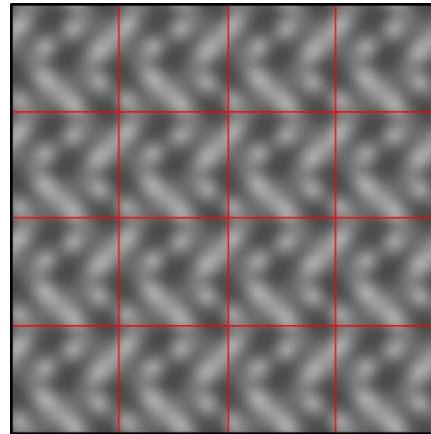
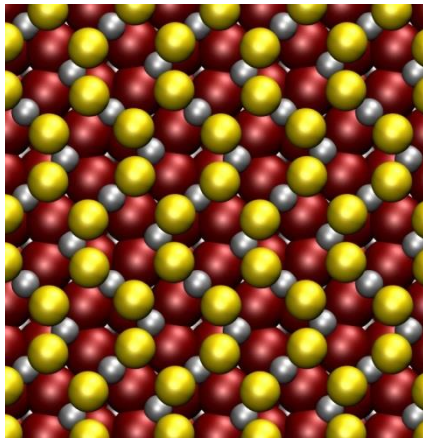
Doping effect



➤ Peroxide splitting highly favorable

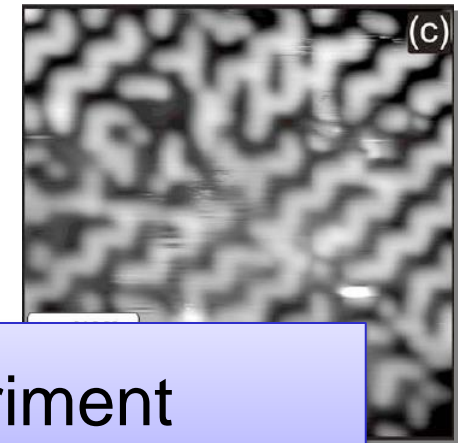
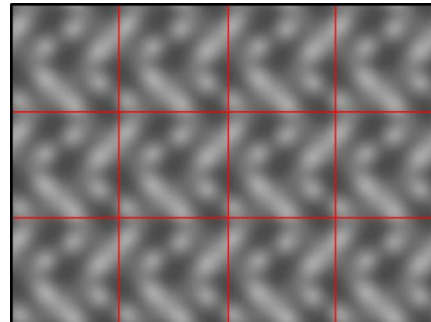
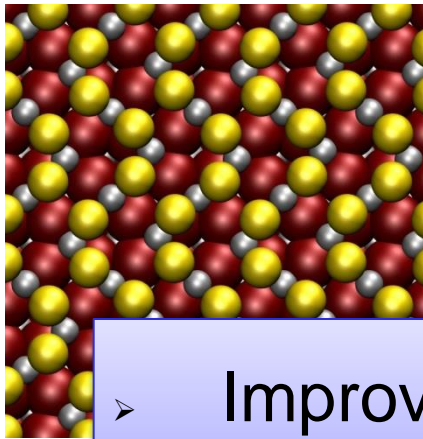
ITO-(001)

Oxygen : ● Indium : ●
Oxygen (1. layer) : ● Indium (1. layer) : ●

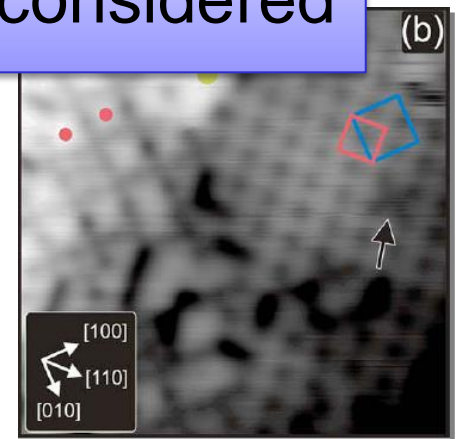
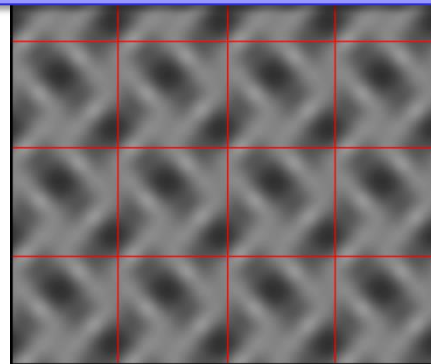
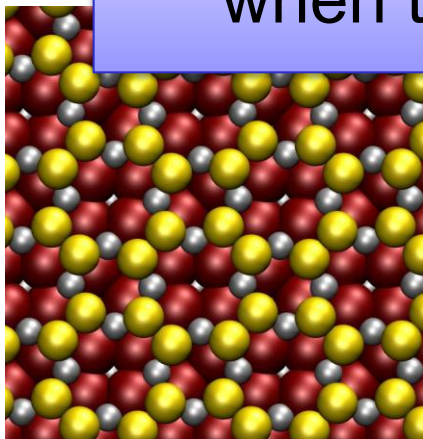


ITO-(001)

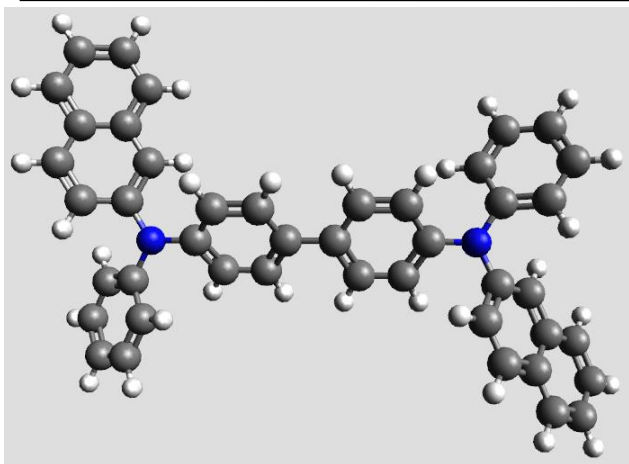
Oxygen : ● Indium : ○
Oxygen (1. layer) : ● Indium (1. layer) : ●



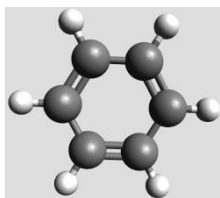
➤ Improved agreement with experiment when the effect of Sn doping is considered



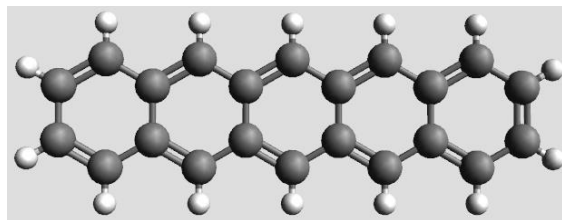
Organic: small molecules (SM) as HTL



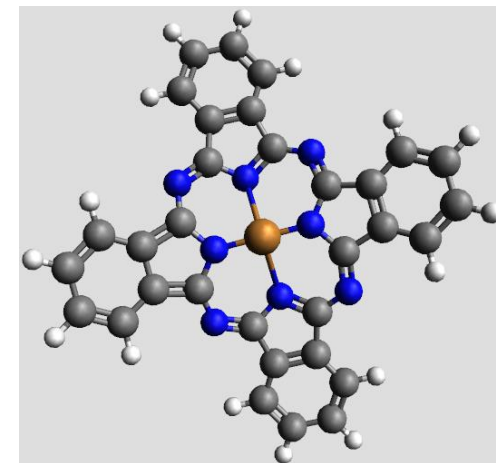
α -NPD
(N,N'-di(1-naphthyl)-N,N'-diphenylbenzidine)



Benzene



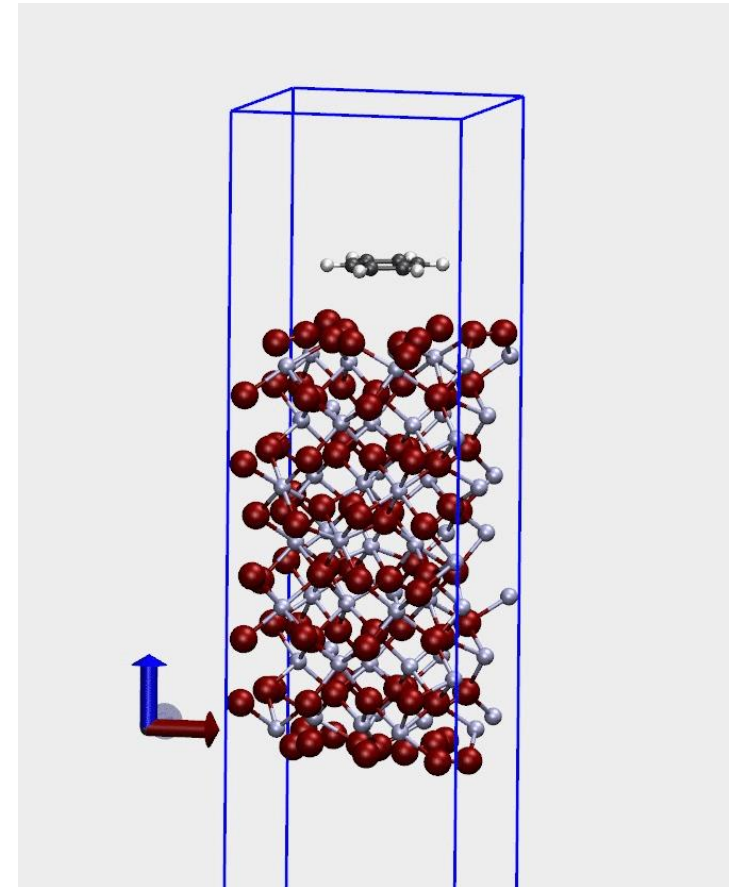
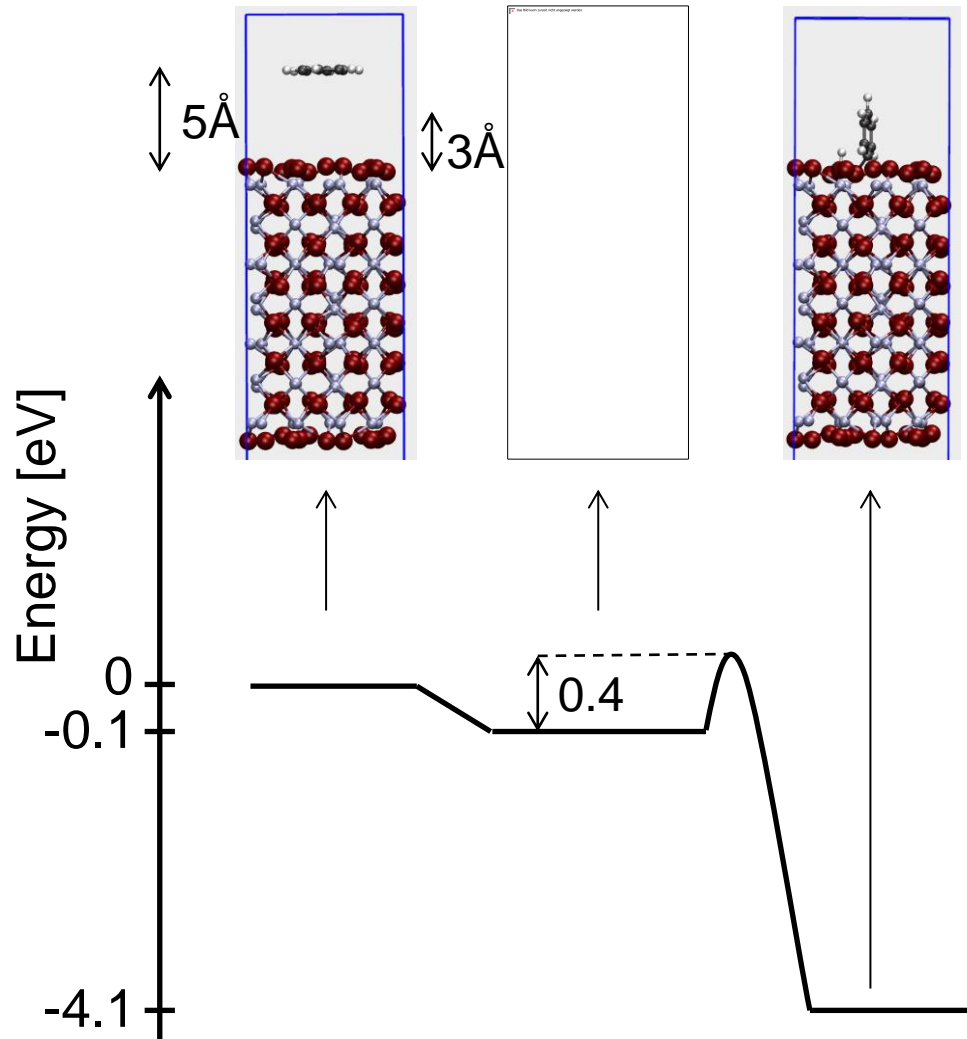
Pentacene



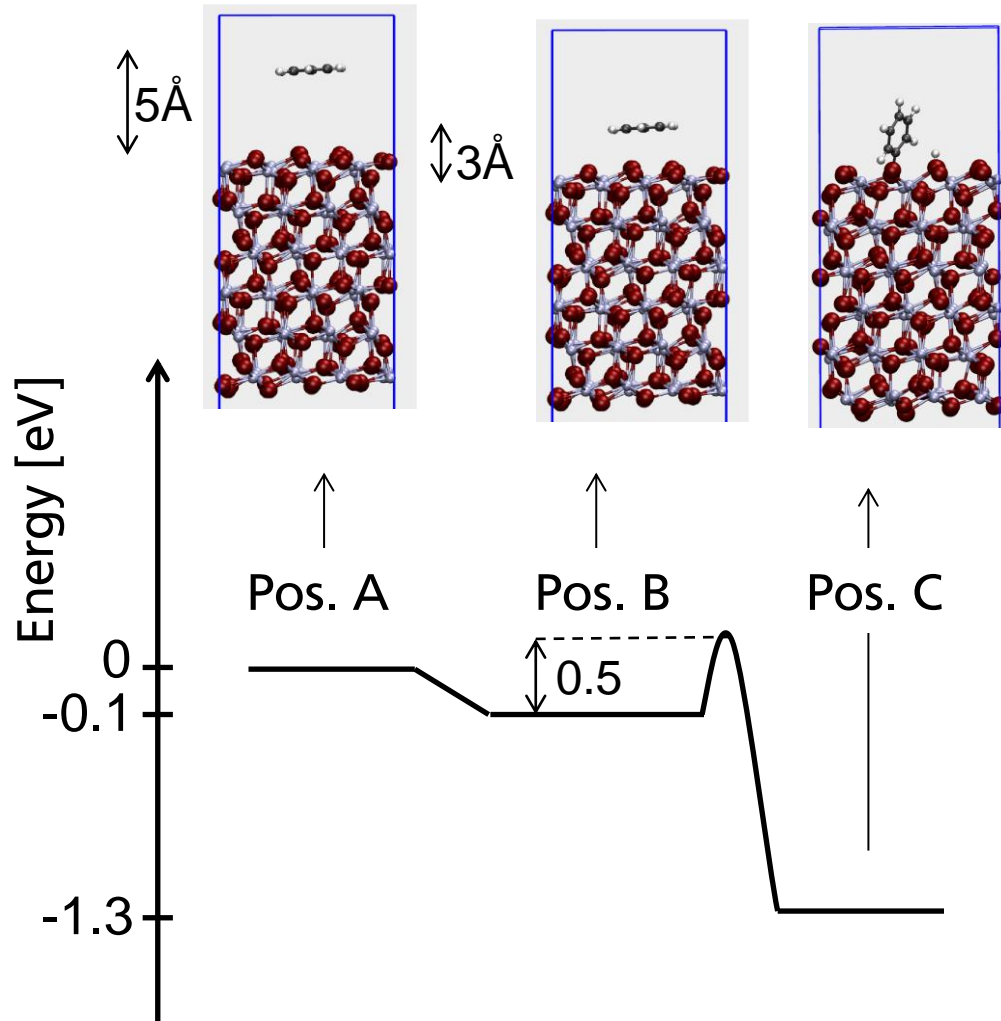
CuPC
(copper phthalocyanine)

- Carbon
- Hydrogen
- Copper
- Nitrogen

Kinetic barrier on In_2O_3 -(100) peroxide surface



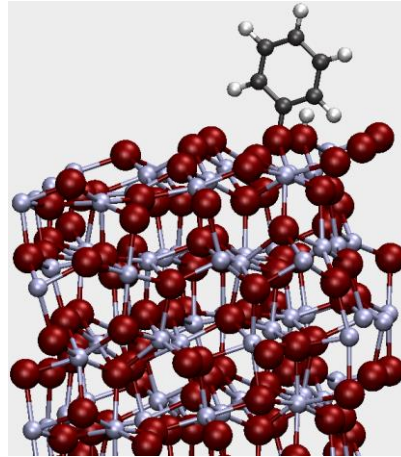
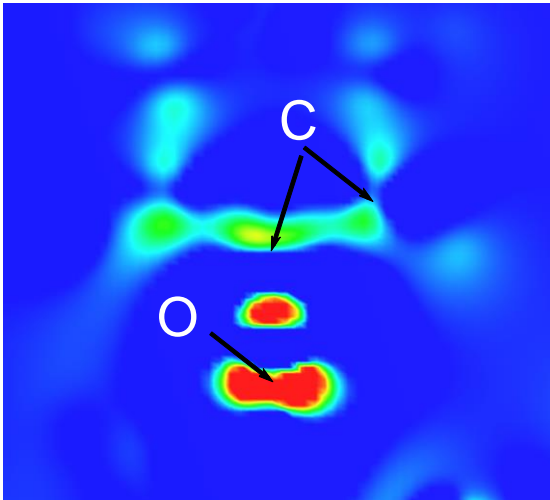
In₂O₃-(111) surface



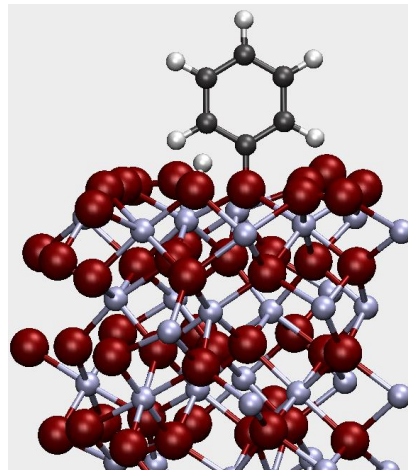
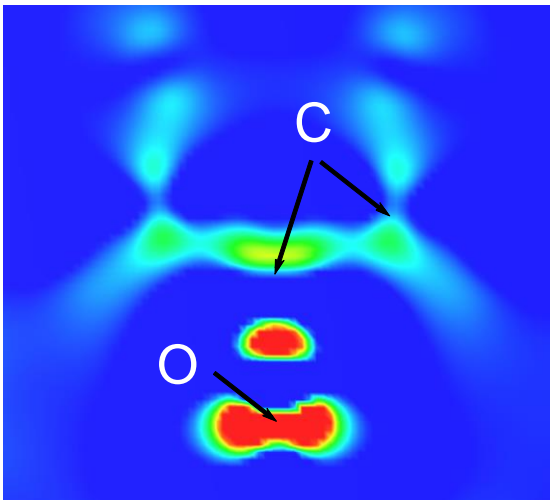
	(100)	(111)
Pos. A	0	0
Pos. B	-0.1	-0.1
Pos. C	-4.1	-1.3
barrier	0.4	0.5

Nature of bonding

(111)



(100)



- The differential charge density shows a covalent bonding between the C (of benzene) and O (of In₂O₃) on (111) and (100) surface.

Conclusions

- Oxygen vacancies are shallow donor defects in In_2O_3 under reducing conditions
- Principal acceptor in In_2O_3 is the negative oxygen interstitial, whereas it is the tin vacancy in SnO_2
- Dopability of SnO_2 is exceeding that of In_2O_3 -> Proper dopands ?
- Oxygen self-diffusion is mediated by oxygen dumbbells and negative oxygen interstitials
- In_2O_3 -(001) surfaces show a propensity towards electron depletion layers
- In polycrystalline ITO films large variations of the ionization potential can be expected
- (001) surfaces of ITO differ from those of pure In_2O_3
- Peroxide surfaces are destabilized by Sn doping; Oxide surfaces are stabilized by high Fermi energy and segregation of Sn
- STM contrast of ITO (001) surfaces can be explained qualitatively only by taking surface charging into account
- Adsorption of benzene favorable at In_2O_3 -(100) and -(111) O-site
- Covalent bonding between C (of benzene) and O (of In_2O_3)